MICROMACHINED PIEZOELECTRIC-ON-SILICON PLATFORM

FOR RESONANT SENSING AND ENERGY HARVESTING

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The Academic Faculty

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Jenna L. Fu

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MICROMACHINED PIEZOELECTRIC-ON-SILICON PLATFORM FOR RESONANT SENSING AND ENERGY HARVESTING

Approved by:

Professor Farrokh Ayazi, Chair, Advisor
School of Electrical and Computer Engineering
Georgia Institute of Technology

Professor Ying Zhang
School of Electrical and Computer Engineering
Georgia Institute of Technology

Professor Mark G. Allen
School of Electrical and Computer Engineering
Georgia Institute of Technology

Professor Michael H. Bergin
Schools of Civil and Environmental Engineering and Earth and Atmospheric Sciences
Georgia Institute of Technology

Professor Oliver Brand
School of Electrical and Computer Engineering
Georgia Institute of Technology

Date Approved: June 25, 2013
To my parents,

Ta-Wei and Hui-Ying Chen Fu
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“Trust in the Lord with all your heart and lean not on your own understanding; in all your ways acknowledge Him, and He will make your paths straight.” - Proverbs 3:5-6
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SUMMARY

This dissertation presents a microelectromechanical systems (MEMS)-based environmental monitoring platform comprising silicon-based, piezoelectrically-actuated resonant sensors and kinetic energy harvesters. All devices are realized using thin-film piezoelectric-on-substrate (TPoS) technology, which provides a path to integrate various functionalities on a single substrate with MEMS components. TPoS resonators, which exhibit high quality factors ($Q_s$) in air, are suitable for sensing when access to the outside world is required. The low-power operation of TPoS oscillators further qualifies such a platform for mobile and portable systems.

To validate the TPoS platform, gravimetric humidity sensing is demonstrated with thermally-corrected output response by an uncoated “reference” temperature sensor. Also presented are AlN-on-Si sensors for toluene and xylene, which are pollutants of great importance for indoor and outdoor air quality as well as health screenings. Silicon dual-mode resonators and oscillators for self-temperature sensing are also explored. Dual-mode thermometry exploits the inherent frequency-temperature dependence of silicon to accurately and locally measure device temperature, forming an essential building block of highly stable oscillators and sensors.

Multi-axis kinetic energy harvesting (KEH) devices with integrated frequency-upconverting transducers are introduced. Devices are micromachined on the same substrate as TPoS resonant sensors and have an individual volume in mm$^3$, enabling applications in wireless autonomous sensor nodes. In remote locations where continuous operation may be required, TPoS energy harvesters can provide battery replacement or recharging alternatives that do not increase overall system size.
1 INTRODUCTION

The world contains an abundance of sophisticated sensors that transform external stimuli into useful information. All living organisms are sensitive to aspects of the external environment, such as light, temperature, vibration, electric fields, and biomolecules. Besides natural sensors, we interact with countless man-made sensors every day. For example, industrial processes require continuous monitoring of temperature, humidity, and other parameters, while today’s automobiles are equipped with sensors that control emissions levels, deploy airbags, and regulate cabin climate. Additionally, smart buildings have sustainable features and sensing systems that conserve energy while maintaining a healthy indoor environment for all users.

The increasing ubiquity of mobile devices in today’s society has increased the demand and potential for miniaturized, integrated sensors. Each generation of smartphone has provided users with enhanced experiences via gaming, navigation, and various location-based services, many of which utilize microelectromechanical systems (MEMS) devices. The versatility of micromachining technology has opened the door to additional functionalities that have yet to be fully explored.

1.1 SENSOR SYSTEM OVERVIEW

A complete sensor system comprises several functional blocks (Figure 1.1), the first of which is typically a filter/pre-conditioner that collects and prepares the chemical, biological, or physical analyte for measurement. This interface with the surrounding environment is coupled to a transducer that transforms the analyte to a measurable signal. Post-transduction data requires mixing or amplifying via conditioning circuitry. Finally, the output module produces a quantity that can be easily interpreted by the user (e.g.
voltage, frequency) [1]. This thesis focuses on the micro-scale transducers that convert the analyte to an electrical signal, while considering the effects of device performance on oscillator interface circuitry used to generate sensor readout.

Sensors are typically characterized by the following metrics: sensitivity, the ratio of change in sensor output to a change in analyte; resolution, the minimum detectable change in analyte quantity at the sensor output; and selectivity, or the capability of distinguishing one analyte from another [2]. Although most sensing applications require high sensitivity, low resolution, and high selectivity, the exact specifications are application-dependent. For example, ultra-stable reference oscillators benefit from the highest sensitivity, lowest resolution thermometers to compensate for thermal effects on frequency, while consumer applications can sacrifice sensitivity or resolution to enable portability and low-power operation. Selectivity, which becomes challenging for environmental monitors, can be addressed by implementing sensor arrays.

Figure 1.1: Schematic diagram of basic sensor operation, where an electrical signal is produced in response to an external stimulus.
1.2 CURRENT MICRO-SENSING TECHNOLOGY

1.2.1 RELATIVE HUMIDITY AND TEMPERATURE (RH&T) SENSORS

Chemical and biological microsensors were predicted to enter the mobile phone market in 2014-2015 via portable health and air quality monitoring applications [3]. While micromachined inertial and pressure sensors have become standard amenities in many of today’s mobile devices, the various challenges associated with environmental sensor integration have yet to be thoroughly addressed. This year, however, marks a milestone for miniaturized environmental monitors: the Samsung Galaxy S4, top-rated by Consumer Reports within a month of its release [4], touted the integration of Sensirion’s latest single-chip relative humidity and temperature (RH&T) sensor among numerous attractive features. A comparison of commercial RH&T micro-sensing technology provides insight into Sensirion’s success cracking the consumer market (Table 1.1).

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<tr>
<td>Sensirion</td>
<td>Bandgap</td>
<td>±0.4</td>
<td>Capacitive</td>
<td>±4.5</td>
<td>2 × 2 × 0.8</td>
<td>Monolithic</td>
<td>$20</td>
</tr>
</tbody>
</table>

Table 1.1: Comparison of commercial RH&T micro-sensors.

Information obtained from digikey.com and official company websites. Unit price range provided for 1 to 1000 parts (if possible) except for Sensirion, which currently only sells parts in quantities of 50,000.

Most RH&T chips employ similar principles of operation: humidity sensing is achieved by a thin-film polymer capacitor that changes dielectric constant with absorbed or
desorbed moisture, while a silicon diode band-gap circuit performs on-chip thermometry (in IST’s device, a platinum resistance temperature detector is used). What truly separates one generation of sensors from the next is the level of system integration. IST sells a combined RH&T sensor but the discrete interface electronics dramatically increase system footprint. GE and Honeywell’s sensors and circuits, although separate entities, are co-located in a single system-on-chip (SoC) package to eliminate performance issues resulting from discrete components. Silicon Laboratories and Sensirion, however, take a monolithic approach to achieve single-chip RH&T solutions small enough to fit in a mobile phone. Sensirion’s technology seamlessly integrates sensors with CMOS analog and digital signal processing circuitry on the same silicon substrate, enabling a low-power, high-performance system that occupies 4× less area than the already-small Silicon Laboratories chip.

Samsung utilizes the RH&T sensor to display optimum comfort and fitness levels for users, which only scratches the surface of beneficial smartphone apps. Temperature and humidity measurement are fundamental to any multi-sensing platform that discerns chemical compounds and bio-molecules with varying complexity, much like an electronic nose (“e-nose”) [5, 6]. Proper tracking of temperature and humidity in conjunction with important target gases would enable smartphones to not only assess local environmental conditions but also provide information specific to each user.

1.2.2 Commercial Portable Environmental Sensors

Although the term “environmental sensing” initially conjures images of meteorological forecasting, assessment of one’s surroundings based on various environmental parameters is in fact required by a wide spectrum of applications. Because
smartphones travel everywhere with their owners, mobile devices provide a unique platform to implement functionalities that benefit individuals and society on a larger scale. Health and safety ratings for outdoor air are currently provided from stationary locations; however, mobile phone monitoring would enable users to obtain local and customized information depending on their sensitivity to certain contaminants.

Various measures have been taken in the last few decades to reduce emissions, but unhealthy air still persists in many areas of the country and around the world. The U.S. Environmental Protection Agency (EPA) defined exposure standards for common air pollutants considered harmful to public health and the environment: carbon monoxide (CO), nitrogen dioxide (NO$_2$), ozone (O$_3$), particulate matter, and lead [7]. Development of a portable system to monitor the gaseous compounds would be particularly beneficial to residents in urban and industrialized locations.

While a person’s choice of activities could be positively influenced by personalized advice about outside air, time spent indoors often exceeds durations of outdoor exposure. According to comprehensive EPA studies [8], volatile organic compounds (VOCs) from both natural and man-made processes have been linked to both acute and chronic adverse health effects. Indoor concentrations are on average 2 to 5 times higher and even up to 10x higher, regardless of location (rural or industrial). Additionally, elevated concentrations can persist in the air long after a compound-containing product is used. Aside from indoor air quality (IAQ) concerns, VOCs can form explosive mixtures in air (at high enough concentrations) and have been identified as precursors to formation of low-level ozone [9], which reduces agricultural yield and
overall outdoor air quality [10]. In conjunction with temperature and humidity, VOC levels also strongly influence indoor air quality [11].

Portable integrated micro-sensors can be used to provide built-in health and safety features. While technology will never substitute for face-to-face interaction with medical professionals, previous costly and invasive screenings may be quickly performed by devices equipped with smart sensors. Recent examples of great interest include detecting compounds in the breath with lung cancer signatures [12] or even public safety risks such as drunk driving, which could be prevented by a “smartphone breathalyzer” capable of determining dangerous levels of ethanol.

While monolithic RH&T sensors have successfully achieved size and power reduction without sacrificing performance, the CMOS-only platform is limited by the lack of flexibility to integrate different sensing methods required for gas and chemical detection. The most common VOC and gas sensing methods in commercial devices are described in the following two sections.

**Semiconductor (Potentiometric) Gas Sensors**

Potentiometric gas sensors, which determine the amount of analyte adsorbed onto a sensitive material by measuring changes in electric potential, are realized using n-type thick films such as tin dioxide (SnO₂) (Figure 1.2a). SnO₂ adsorbs negatively-charged oxygen atoms onto the surface when heated, forming a potential barrier against current flow and generating a high resistance $R_s$ in fresh air. When reducing gases adsorb onto the sensor surface, the potential barrier—and therefore $R_s$—is decreased significantly (Figure 1.2b) [13]. The target gas concentration $C$ is determined using

$$R_s = A[C]^{-\alpha},$$  

(1)
where $A$ is a constant and $\alpha$ is given by the slope of the $R_s$ curve. Potentiometric sensors have had commercial success for more than 30 years because of their simple thick-film fabrication process and relatively long lifetimes. However, varying temperature and humidity conditions as well as ambient oxygen concentration can affect sensitivity characteristics. To maintain the sensing substrate at a constant temperature, solid-state sensors are equipped with on-chip heaters that require additional power consumption. Potentiometric sensors can have poor selectivity because any reducing gas in the environment can affect the measured conductivity of a metal oxide sensing film.

**Non-dispersive Infrared (NDIR) Gas Sensors**

NDIR is commonly used for carbon dioxide (CO$_2$) and certain hydrocarbons that are difficult to detect by other methods. NDIR addresses the selectivity and sensitivity problems with potentiometric sensing by utilizing the specific IR absorption properties of gases. A typical NDIR sensor consists of an emitter source that generates infrared (IR) radiation, a sample cell or chamber through which the gas is directed, and an IR detector at the opposite end to measure the amount of light absorbed by the gas (Figure 1.3) [14]. Gas concentration $P$ is determined using Beer’s Law.
\[ I = I_o e^{kp}, \]  

where \( I \) is the light intensity measured by the detector, \( I_o \) is the “reference” light intensity in an empty sample chamber, and \( k \) is a system-dependent constant. Any wavelengths not absorbed by the target gas are filtered out before light reaches the detector, enabling highly sensitive and selective detection. However, temperature and humidity variations may still create cross-sensitivity issues in NDIR sensors. Additionally, it is costly and difficult to scale NDIR optics for integration purposes. The only MEMS-based NDIR sensor components fabricated to-date (ICx Photonics, Inc. [15]) are not yet commercially available.

Figure 1.3: Schematic diagram of a CO\(_2\) NDIR sensor.
2 RESONANT MICROSENSORS

Based on the current state of the art in commercial micro-sensing technology, scalability, environmental sensitivity, and integration potential are the main bottlenecks in realizing mobile-friendly systems for trace gas and chemical detection. Devices should be inexpensive to manufacture, yet maintain robust performance with scaling. As illustrated by successful commercial sensors (e.g. Sensirion), the sensing element must ultimately be combined with electronics and signal processing units through monolithic or CMOS-compatible integration. Resonant micro-devices, which provide a frequency output that can be digitally measured with high accuracy [16], meet the aforementioned requirements in terms of integration potential and are strong candidates for implementation of a miniaturized multi-sensing platform.

2.1 GRAVIMETRIC RESONATORS

The model for mechanical resonance in a basic mass-spring system with spring constant \( k \), mass \( m \), and no damping is given by (3), where \( f_0 \) represents the system natural frequency:

\[
f_0 = \frac{1}{2\pi} \sqrt{\frac{k}{m}}. \tag{3}
\]

Thus, modulation of resonator properties such as stiffness or mass will induce a frequency shift. Provided that mass change dominates over stiffness change, \( \Delta f \) due to mass loading can be approximated by

\[
\Delta f \approx -\frac{1}{2} f_0 \frac{\Delta m}{m}, \tag{4}
\]
where $\Delta m/m$ is the relative mass change ($\Delta m/m \gg \Delta k/k$). Coating the resonator with a material possessing high affinity for the target analyte facilitates mass loading based on a chemical reaction or physical sorption, resulting in a frequency downshift [17]. Sensor frequency can be tracked by interfacing the resonator with electronics to form an oscillation loop. The resulting output signal with frequency $f_0$ can be measured in real time and correlated to the sensor response.

Reducing the necessary circuit drive levels to lock into oscillations requires a resonator with low motional resistance [18], while frequency stability depends on quality factor $Q$, which quantifies “sharpness” of the resonance peak. Electrically, $Q$ relates $f_0$ to the resonator bandwidth (smaller bandwidth indicates higher $Q$). Mechanically, $Q$ is defined by energy stored versus energy lost per cycle, or (5)

$$Q = 2\pi \frac{\text{energy stored}}{\text{energy dissipated per cycle}}. \quad (5)$$

Because signal amplification naturally occurs by a factor of $Q$ at resonance, small amplitudes can be increased without requiring significant power consumption from amplifying electronics. Furthermore, mechanical resonators achieve $Q$s in the thousands and even millions [19] compared to electrical LC resonant circuits exhibiting $Q$s in the tens [20]. In sensor applications, high-$Q$ operation is essential because any broadening of the oscillator frequency spectrum translates to phase-noise (jitter in time domain), which ultimately leads to uncertainty in sensor readout.
2.2 RESONATOR TRANSDUCTION METHODS

This section discusses three common integrated transduction methods for resonators and why piezoelectric actuation is the most promising candidate for gravimetric micro-sensing.

2.2.1 CAPACITIVE AND THERMAL ACTUATION

Capacitive resonators (Figure 2.1a) are excited into resonance when an AC signal is applied to conductive electrodes separated by a capacitive gap, creating a fluctuating electrostatic force [21]. Capacitively-actuated devices typically have very high $Q$ values [22] because the transducers are physically separated from the resonating body, reducing energy loss via actuation and sensing mechanisms [21]. As a result, such devices typically require a large DC polarization voltage ($V_p$) for operation and also have higher motional resistance, which creates difficulties for electronics interfacing.

Alternatively, resonators can be thermally actuated (Figure 2.1b) by passing a combination of DC and AC current through a structure containing regions of varying electrical resistance. Fluctuating temperature gradients (and therefore, fluctuating thermal expansion) generates sufficient force to excite various extensional resonance modes and has been demonstrated at frequencies as high as 61 MHz [23]. However, the AC and DC components required for thermal actuation also leads to high power consumption; additionally, sensor designs must be thoroughly optimized to prevent adverse effects on the sensitive coating—especially if it is polymer-based—due to high localized temperatures.
2.2.2 PIEZOELECTRIC TRANSDUCTION

While capacitive and thermal actuation enable high-performance resonators via CMOS-compatible processes, both transduction methods present challenges for integrated microsensors. Piezoelectric transduction, utilized in numerous macro-scale applications, has also been employed in micromachined resonators and sensors.

The piezoelectric effect is the conversion of mechanical strain to electric charge that occurs in materials that lack a center of symmetry. Piezoelectric materials also exhibit the inverse piezoelectric effect, in which mechanical deformation results from the application of electric charge. Since its discovery in the late 1800s, the bi-directional behavior of piezoelectricity has been exploited for various sensors and actuators, resonant devices in particular. Piezoelectrically-driven resonators (Figure 2.1c) have several advantages including low motional impedance [26] and the ability to operate without DC voltages. These characteristics make piezo devices desirable for interfacing with electronics and low-power operation compared to capacitively- or thermally-actuated resonators.

![Silicon resonators utilizing (a) capacitive, (b) thermal, and (c) piezoelectric actuation.](image)

Figure 2.1: Silicon resonators utilizing (a) capacitive [22], (b) thermal [24], and (c) piezoelectric actuation [25].
2.3 PIEZOELECTRIC ACOUSTIC WAVE MICRO-DEVICES

2.3.1 PIEZOELECTRIC MATERIALS

Quartz, a strong crystalline piezoelectric material, has been successfully manufactured for high-precision frequency references, film thickness monitors, and chemical sensors in the past few decades [17, 27-29]. However, the bulk mode frequency of quartz resonators is inversely related to film thickness, which pre-determines the operation frequency for an entire quartz substrate. Various challenges associated with quartz and CMOS integration ignited research and development of micromachined alternatives for timing and sensing, as well as suitable piezoelectric material synthesis methods for such micro-devices. Piezoelectric polymers such as PVDF [30] and polycrystalline thin films including lead zirconate titanate (PZT), zinc oxide (ZnO), and aluminum nitride (AlN) can be realized through sputtering, sol-gel deposition [31], and more recently, inkjet printing [32].

2.3.2 PIEZOELECTRIC MICRO-RESONATORS

Advancements in thin film deposition and fabrication techniques contributed to recent development in piezoelectrically-actuated acoustic wave devices, including thin film bulk acoustic resonators (FBARs) [33], surface acoustic wave (SAW) resonators, and flexural micro-cantilevers.

FBARs comprise a piezoelectric thin film material stack supported over an air cavity or, in solidly-mounted resonator (SMR) configuration, reflecting layers to provide acoustic isolation from the substrate. Devices typically resonate in a thickness-extensional mode but can be designed to excite a thickness-shear mode, as in quartz (Figure 2.2a). FBARs can be batch-fabricated with hundreds to thousands of devices on a
single substrate, achieving high $Q$ and mass sensitivity for various applications into the GHz regime [34]. However, resonant sensors with thickness-dependent frequency are prone to $Q$ degradation since functional coatings are applied in the direct path of the acoustic wave.

SAW resonators exploit a mode with maximum wave propagation in the surface of a piezoelectric material [1] (Figure 2.2b). SAW resonators have been used in various signal processing applications, but the high sensitivity to surface perturbations has also made them attractive for sensing. Because SAW devices are based on mono-crystalline piezoelectric materials such as lithium tantalate (LiTaO$_3$), lithium niobate (LiNbO$_3$), or quartz [35], they are incompatible with standard CMOS electronics. Furthermore, devices are not easily miniaturized and typically have dimensions in the mm-range [36].

While in-plane resonance modes have been used to implement cantilever-based chemical sensors [37], micro-cantilevers are typically excited in a flexural, out-of-plane mode (Figure 2.2c). Cantilever resonators often require lasers or external equipment to measure deflection, which further complicates miniaturization and integration.

Figure 2.2: Piezoelectric resonant sensors based on (a) bulk/thickness shear (quartz and FBAR with piezoelectric material sandwiched between metal electrodes), (b) surface (SAW device with interdigitated electrodes on a piezoelectric substrate), and (c) flexural (cantilevers with piezoelectric stack deposited on a structural material).
2.4 SENSOR IMPLEMENTATION CHALLENGES

As demonstrated by piezoelectric acoustic wave micro-devices, resonators can be rendered sensitive to physical and chemical quantities via functional coatings, showing that gravimetric resonant detection is a promising alternative to traditional techniques for miniaturized sensors. However, portable, multi-sensing systems require flexibility to accommodate for diverse materials used in sensor functionalization while maintaining low power and low cost.
3 THIN-FILM PIEZOELECTRIC-ON-SILICON MICRO-
PLATFORM

Electrostatically-transduced (capacitive) resonators, which typically require vacuum encapsulation to achieve high $Q$, are not suitable for sensing applications that require exposure to the environment. Furthermore, their high motional resistance and the need for a DC polarization voltage complicate oscillator interfacing and increase system power consumption. Thermally-actuated resonators, which also require DC voltages to operate, are power-hungry and susceptible to reduced analyte sensitivity as a result of sorption layer heating [38]. In contrast, low-power sensing is achievable using thin-film piezoelectric-on-substrate (TPoS) technology [25], which combines efficient piezoelectric transduction requiring no DC voltages with a high-$Q$ structural material such as nanocrystalline diamond or silicon [26, 39]. Single-crystal silicon, which has emerged as a viable alternative to quartz for timing and sensing as a result of its excellent mechanical and acoustic properties, is chosen as the substrate in this work to enable cost-effective fabrication and compatibility with CMOS electronics.

An integrated micro-platform consisting of TPoS gravimetric gas sensors and kinetic energy harvesters is proposed for micro-sensing system implementation (Figure 3.1).

3.1 PLATFORM OVERVIEW

The TPoS gravimetric sensor array can be selectively functionalized to detect key chemical and biological markers in the environment. The maturity of silicon micromachining technology enables a material with desirable electrical and mechanical properties to be exploited for multi-functional integrated systems. Unlike quartz, silicon
resonators are not natively piezoelectric and suffer from large temperature-induced frequency shifts via modulation of elastic properties, while the combined effects of temperature and humidity influence chemical reactions and sorption in sensitive layers. Since sensitivity is directly tied to $f_0$, modes with lithographically-definable frequencies enable array implementation on a single wafer to accommodate multiple levels of sensitivity.

Besides mobile and consumer applications, the TPoS platform can also be applied to self-powered wireless data transmission for autonomous sensor networks [40]. Sensors in remote or dangerous locations may need to operate for several years without interruption and require alternatives to battery replacement or recharging that do not increase overall system size. Thus, it would be beneficial to directly integrate kinetic energy harvesting devices with the proposed platform.

Figure 3.1: Schematic diagram of integrated micro-platform consisting of TPoS gravimetric resonant sensors and kinetic energy harvesting (KEH) devices.
3.2 FABRICATION PROCESS FLOW

The TPoS fabrication process flow [41] was customized to facilitate integration of sensors and energy harvesting devices on a single substrate. All steps except AlN deposition were performed in the Georgia Tech Pettit and Marcus cleanroom facilities.

3.2.1 CHOICE OF PIEZOELECTRIC MATERIAL

Sol-gel deposited films such as PZT require poling, which subjects the film to a strong electric field for an extended period of time, to obtain the required piezoelectric crystal structure. On the other hand, piezoelectric zinc oxide (ZnO) and aluminum nitride (AlN) can be deposited by low-temperature RF magnetron sputtering, which may be necessary for various sensor functionalization processes. Piezoelectrically-transduced sensors based on both ZnO and AlN have been investigated in this work, and advantages and challenges of sensor integration are briefly discussed.

ZnO, which exhibits high electromechanical coupling, is readily available in the Georgia Tech cleanroom and has been optimized to produce RF-sputtered films with high c-axis orientation, confirmed by x-ray diffraction (XRD) analysis on actual process wafers. E-beam evaporated gold, which has been previously established as a suitable electrode material for ZnO [25], serves as a useful bio-functionalization layer for self-assembly processes and is also immune to oxidation from exposure to ambient air. However, a major drawback of ZnO is its high reactivity to common wet and dry etchants such as HF, which imposes restrictions on post-deposition fabrication steps. Over time, the reactivity of ZnO in the environment could result in degraded resonator performance. Furthermore, the risk of CMOS contamination by Zn ions eliminates the possibility of monolithic integration and introduces packaging complications in 3D stacked CMOS and
MEMS systems. Finally, without strict stoichiometric control of the deposited film, ZnO can be fairly susceptible to DC leakage and requires additional isolation layers that may also negatively impact sensor performance [37].

AlN, which possesses an identical crystal structure (wurtzite) and similar transverse piezoelectric coefficient to ZnO, is CMOS-compatible and less susceptible to DC leakage than ZnO [42]. Since AlN exhibits lower reactivity to many common wet and dry etchants, AlN-on-Si devices provide additional flexibility for array implementation and post-fabrication sensor functionalization.

### 3.2.2 ZNO-ON-SI FABRICATION PROCESS FLOW

ZnO-on-Si resonators were fabricated on 5-µm thick, high-resistivity silicon-on-insulator (SOI) wafers (Figure 3.2a-f). First, 0.05 - 0.1 µm of gold is electron-beam evaporated and patterned by liftoff, removing Shipley SC1827 positive photoresist with acetone to form the bottom electrode (Figure 3.2a). To improve adhesion to the Si device layer as well as the piezoelectric film, a thin layer of e-beam evaporated titanium sandwiches the gold layer.

Approximately 0.7 µm ZnO is deposited by RF sputtering with 125 W RF power and substrate temperature of 210 °C for 100 minutes. Parameters consistently used in each ZnO deposition are listed in Table 3.1. After each deposition, c-axis orientation required for piezoelectric coupling is confirmed through XRD analysis. Rocking curve analysis (ω scan) revealed typical full-width-half-maximum (FWHM) values of about 2.2°, indicating a narrow distribution around the surface normal and good crystalline quality (Figure 3.3).
The ZnO film is subsequently patterned in acetic acid:deionized water (1 mL:150 mL ratio achieves optimal pH of 3) to remove ZnO from all areas of the wafer except the active resonator body (Figure 3.2b). 0.1 µm of gold is evaporated and patterned by liftoff to create the top electrode and complete the piezoelectric stack (Figure 3.2c). To define resonator supports and the overall shape by forming trenches in the silicon device layer, deep reactive ion etching (DRIE) is performed in an STS ICP system (Figure 3.2d). The device is released by etching the handle layer from the backside with SF₆ (Bosch process, STS ICP) and subsequently removing the buried oxide (BOX) layer from the backside with a C₄F₈ mixture using a Plasma-Therm ICP etcher (Figure 3.2e). More details on sensor functionalization, which is performed after device release (Figure 3.2f), are provided in chapter 4.

![Figure 3.2: Scanning electron microscope (SEM) image of released ZnO-on-Si resonator and accompanying process flow.](image-url)
Table 3.1: Parameters used for ZnO RF sputter deposition in the Georgia Tech IEN cleanroom.

<table>
<thead>
<tr>
<th>Initial pumpdown pressure</th>
<th>Process pressure</th>
<th>RF power</th>
<th>Ar/O₂ flow ratio</th>
<th>Substrate temperature</th>
<th>Target-to-sample distance</th>
</tr>
</thead>
<tbody>
<tr>
<td>4 to 6 µTorr</td>
<td>6 mTorr</td>
<td>125 W</td>
<td>1:1</td>
<td>210°C</td>
<td>9.5 cm</td>
</tr>
</tbody>
</table>

Process assumes ZnO target used in argon (Ar) and oxygen (O₂) ambient.

Figure 3.3: Typical XRD rocking curve (ω) scan to confirm c-axis orientation of ZnO sputtered in the Georgia Tech Pettit cleanroom facility.
3.2.3 **ALN-on-SI Fabrication Process Flow**

This section outlines the basic AlN-on-Si process flow for resonators and sensors, fabricated on silicon device layers ranging from 5 to 20 µm. Modifications to facilitate energy harvester integration are described in section 6.4.

Molybdenum electrodes sandwiching a 0.7 – 1 um thick RF-sputtered AlN layer are deposited on blank SOI wafers by the OEM Group, Inc. and shipped to Georgia Tech for remaining process steps. Typical FWHM of the deposited AlN films, as measured by OEM, ranges from 1.63 to 2°. Because piezoelectric stack layers are successively deposited, fabrication begins with patterning the top Mo electrodes in SF$_6$ plasma using a Shipley SC1813 photoresist mask (Figure 3.4a). Next, PECVD oxide is deposited on both sides of the wafer (~300 nm topside, ~5 µm backside). The topside thin oxide is protected with photoresist while the backside oxide is BOE-etched to define device release cavities. The top oxide is patterned with an SC1827 photoresist mask, and AlN is subsequently etched in TMAH heated at 40°C to expose ground pads in the bottom Mo layer (Figure 3.4b). Additional oxide is deposited on the wafer topside and patterned with SC1827; a Plasma-Therm ICP tool is used to etch oxide followed by AlN and bottom Mo. The device layer Si is etched in an STS ICP (low-frequency Bosch recipe) to define the resonator geometry (Figure 3.4c). Finally, resonators are released from the wafer backside using a high-frequency Bosch etching recipe (STS ICP) to remove the handle Si and buried oxide layers (Figure 3.4d). As with the ZnO-on-Si resonant sensors, released devices are functionalized using the Bioforce Nano eNabler (discussed in chapter 4, section 4.3.2).
Figure 3.4: Fabrication process flow for AlN-on-Si resonant sensors: (a) top Mo patterning; (b) backside oxide mask patterning and AlN etch to expose bottom Mo; (c) device Si trench definition; and (d) handle Si and BOX layer removal.
3.3 MATERIALS FOR SORPTION-BASED MICRO-SENSING

TPoS resonators function as thermometers based on the intrinsic temperature coefficient of frequency (TCF) of silicon. Sensitizing resonators to a particular gas-phase analyte requires a functional coating that facilitates sorption or a chemical reaction, resulting in mass loading that can be detected by frequency shift. Gravimetric micro-resonators have employed a variety of materials for gas sensing; however, application-dependent factors such as reversibility, repeatability, and response time of analyte reaction should be considered in material selection.

3.3.1 POROUS MATERIALS AND FRAMEWORKS

Adsorbate materials such as carbon and zeolites have been used for years by environmental scientists to remove harmful pollutants from the air. Naturally-porous zeolites, or “molecular sieves,” contain an alumino-silicate crystal structure with uniform nanometer-sized pores determined by silicon-to-aluminum ratio [43]. The Si/Al ratio, which controls the hydrophobicity, defines the analytes to which a particular zeolite structure will have the greatest affinity (non-polar to hydrophobic, polar to hydrophilic). The tunable characteristics of zeolites have been exploited for selective VOC and gas detection using quartz crystal microbalances (QCMs) [44, 45], micro-cantilevers [46], and semiconductor gas sensors [47, 48]. Zeolite-modified sensors exhibited enhanced responses, most likely as a result of increased surface area on which molecules can adsorb. Since zeolites are grown by high-temperature hydrothermal processes, most sensors were functionalized spin-coating, dropping, or evaporating a suspension of crystals [46]. However, uniformity of the final deposited films typically varies significantly from one sensor to the next, which is unacceptable for batch-processing.
Metal-organic frameworks (MOFs) are nano-porous materials with crystalline structure defined by rigid organic molecules linked to metal ion clusters. Various micro-sensors modified with MOFs [49-51] have been used to detect humidity and various gases. However, like zeolites, the MOF functionalization process requires immersion or deposition of a suspension, which creates challenges in achieving uniformity control.

Alumina- and silicon-based porous films have also been utilized for humidity sensing [52, 53]. However, the sensor fabrication process exposes the device to large current density, harsh chemicals, and temperatures in hundreds of degrees Celsius, which is not compatible with CMOS integration and ultimately degrades sensor performance.

Overall, porous materials and frameworks exhibit enhanced analyte adsorption and selectivity while remaining chemically and thermally stable. However, slow diffusion through these materials results in extremely long sensor response times, with certain sensors failing to equilibrate after several hours of constant exposure [49]. Even if the sensor responds rapidly [46], full recovery (complete analyte desorption) requires the material to be heated, consuming excess power that cannot be afforded in miniaturized sensing systems.

3.3.2 POLYMERIC MATERIALS

Polymers are advantageous for real-time sensor array applications where faster response time is required. Adsorption-based reactions are completely reversible and unlike hard, porous materials, do not require thermal desorption to “reset” the sensor. From an integration standpoint, polymeric materials have been applied to sorption-based gravimetric sensing [17, 54-56] because uniform thin films are easily achieved by spin-coating, a repeatable batch-fabrication process. Furthermore, recent studies have shown
that sorptive polymers can be tailored for sensitivity and selectivity by adding functional
groups and modifying other properties in synthesis [57].

Polymers are not without drawbacks, however, as they are inappropriate for high-
temperature environments and have limited lifetime compared to hard materials because
of their lower stability. However, polymeric materials are still strong candidates for
sensing in consumer applications and even industrial applications where temperature is
not a concern.

Although versatile options to modify the sensitivity of micromachined silicon
resonators for VOC and gas detection are available, polymers for sorption-based sensing
have been characterized in this dissertation. Details of gravimetric sensor
functionalization are provided in chapter 4.
4 TPOS GRAVIMETRIC RESONANT SENSORS

This chapter introduces piezoelectric-on-silicon resonant sensors with annexed platforms. Design principles and tradeoffs between mass sensitivity and resonator performance are discussed.

Section 4.2 presents micromachined temperature and relative humidity (RH) sensors using ZnO-on-silicon, lateral-mode bulk acoustic wave (BAW) resonators. An uncoated reference resonator is utilized in conjunction with an identical, humidity-functionalized resonator to yield a thermally-compensated RH sensor. The reference sensor exploits the dual-mode response of first and third-order in-plane extensional modes to eliminate RH sensor dependence on temperature. Polyethyleneimine (PEI) is deposited on annexed platforms of the BAW resonator, enabling operation from 20 to 92 %RH with high $Q$ in air. A sensitivity of 1.2 kHz/%RH from 20 to 60 %RH and 14 kHz/%RH from 60 to 92 %RH is measured at room temperature. The RH sensor combined with the reference sensor occupies a total area of 0.053 mm$^2$.

The final section in this chapter presents aluminum nitride (AlN) thin-film piezoelectric-on-silicon (TPoS) resonators optimized for gravimetric gas sensing. Sensors with polyisobutylene (PIB)-functionalized annexed platforms exhibit a linear sensitivity of -0.09 Hz/ppm to toluene and -0.18 Hz/ppm to xylene. The annexed platform design enables the sensor to maintain a high $Q$ of 3700 in air and low motional resistance of 1.1 kΩ during operation, which ensures low-power oscillator interfacing and readout. AlN-on-Si resonators with annexed platforms are suitable for array implementation in environmental sensing micro-systems that measure temperature, humidity, and various gases of importance to health and safety.
4.1 TPOS RESONATOR WITH ANNEXED PLATFORMS

Ideally, gravimetric sensor frequency should only be affected by changes in mass based on sorption in the functional material. In reality, various factors dependent on sensor structure and materials can contribute to undesired frequency error (e.g. $Q$, temperature, stiffness changes, etc.). Therefore, both enhancements in mass sensitivity and device performance should be considered when designing resonant sensors.

Sensors consist of a centrally-clamped bulk acoustic wave (BAW) resonator, which exhibits an in-plane length-extensional (LE) resonance mode at frequency $f_0$ defined by length $L$ (6)

$$f_0 \propto \frac{1}{2L} \sqrt{\frac{E}{\rho}},$$

with Young’s modulus $E$ of the structural material and density $\rho$. Sensing material is deposited on platforms connected to the central structure by support beams (Figure 4.1). In the simplest case, $W$ and $L_p$ are assigned equal values such that the sensor resembles a rectangular block; however, $W$ can be increased to fine-tune sensor frequency and reduce motional resistance.

![Diagram of AlN-on-Si resonant sensor with annexed platforms.](image)

**Figure 4.1:** Diagram of AlN-on-Si resonant sensor with annexed platforms.
In the sensor mode of operation, platform motion is in phase with the central block extension (Figure 4.2a). In a typical LE resonator (Figure 4.2b), strain varies along $L$ and leads to undesired equivalent stiffness changes due to analyte sorption [58]. In contrast, maximum strain occurs in the central block and beams of the modified sensor (Figure 4.2b). The uniform strain distribution in the platforms enables a predominantly translational motion, minimizing effects on $\Delta k$ and ensuring that adsorption anywhere on the platforms (except near the beams) yields predictable influences on equivalent mass. Thus, mass loading on the platforms (at the maximum vibration amplitude) contributes most significantly to resonance frequency shift. Besides decoupling modal mass and stiffness effects on frequency, the beams confine longitudinal wave propagation to the central structure where signal actuation and detection occur. Thus, damping introduced by analyte sorption on sensing areas does not interfere with resonator acoustic operation. As a result, the sensor maintains high $Q$ in air, which ultimately impacts frequency stability and minimum limit of detection [38].

Figure 4.2: (a) ANSYS mode shape of gravimetric sensor with annexed platforms; (b) comparison of relative modal displacement between conventional LE sensor and sensor with platforms.
4.2 ZNO-ON-SI RESONANT TEMPERATURE AND HUMIDITY SENSOR

4.2.1 TEMPERATURE AND HUMIDITY EFFECTS ON RESONANCE FREQUENCY

Frequency variations caused by the negative TCF of silicon have impeded progress in commercializing such devices. In timing-based applications, temperature-induced frequency variations are passively compensated through oxide deposition [59], p-n doping [60], or by engineering the device geometry [61]. With sensing applications, the focus shifts from eliminating resonator TCF to measuring the temperature and properly accounting for thermal effects on different target analytes in the sensor readout.

While temperature is a key contributor to undesired frequency change in silicon resonant sensors, moisture in the surrounding environment can also affect the sensor output. It is useful to consider relative humidity (RH)—the ratio of actual water vapor pressure ($p_{H2O}$) to the saturation vapor pressure ($p^*_{H2O}$)—in most applications [62]. The dependence of $p_{H2O}$ and $p^*_{H2O}$ on temperature creates difficulty in accurately measuring RH without an accompanying temperature reading.

To separate temperature effects from the desired sensor response, external thermometers are often used to obtain the local temperature, which leads to inaccuracy as a result of thermal gradients between the thermometer and sensor. Sensors with integrated temperature detectors, dual-resonator temperature measurement, and dual-oscillator systems have been developed to address this challenge [63-65]. However, even integrated thermometers increase the overall system size; additionally, measurement inaccuracies arise when temperature information is extracted from multiple devices, as in the dual-resonator and dual-oscillator system. Rather than consuming additional power or area to control the device temperature [66], the resonator itself can be used as a
thermometer to account for temperature-induced frequency changes through a technique known as self-temperature sensing, where the total frequency change of a resonant sensor is modeled by the sum of both temperature- and analyte-induced frequency changes with negligible contributions from other physical parameters [16].

4.2.2 Resonant Temperature and Humidity Sensing

Two micromachined ZnO-on-silicon, length-extensional (LE) mode BAW resonators are used to implement a resonant temperature and humidity sensor. An uncoated resonator serves as a reference temperature sensor, while an identical resonator coated with moisture-sensitive material measures humidity.

Multi-mode excitation in piezoelectric resonators is often troublesome because spurious peaks can interfere with the main mode operation. However, dual-mode self-temperature sensing exploits this characteristic by utilizing the TCF of two modes in a single device to determine resonator temperature [16]. In this case, the first and third length extensional resonance modes ($f_1$ and $f_3$) are actuated and detected by an interdigitated electrode configuration (outlined on the device in Figure 4.3). Any temperature-induced frequency shift in the RH sensor is compensated by extracting local temperature from the reference sensor characteristic to determine the correct value of RH.

![Figure 4.3: ANSYS mode shapes of ZnO-on-Si sensors at (a) $f_1$ and (b) $f_3$; $L = 120 \mu m$, $W = 120 \mu m$.](image)
As shown in Figure 4.3, RH sensing mode $f_i$ exhibits maximum displacement at the block edges rather than in the platforms. In contrast to the extensional sensing mode described in section 4.1, platforms resonate out of phase with the central structure and are therefore expected to have less effect on mass sensitivity than the lower-frequency in-phase mode (with optimal mass sensitivity), which was not detected in this design.

### 4.2.3 Functionalization Process for Humidity Sensing

Spray-coating [37], spin-coating, complete immersion [67], and inkjet deposition [68] have been used to functionalize sensor surfaces for various target molecules. The first three techniques are compatible with batch processing but require a mask or stencil to selectively pattern certain areas on a wafer. Although certain polymers have been defined by photolithography [57], water-soluble polymers would be dissolved in standard developer solutions and are therefore not suitable for lithographic patterning. Typical shadow mask feature size is too large unless the mask is placed in direct contact with the wafer, which can damage devices when removed after functionalization. Inkjet deposition does not require additional masks, but extensive characterization is required to achieve a uniform spot size with desired properties. Furthermore, because the nozzle is placed a

![Figure 4.4](image_url)

**Figure 4.4:** (a) Patterns deposited using a Jetlab II tool. Colored rings indicate thickness variations as confirmed by Wyko optical profilometer measurement (inset); (b) patterns deposited with the Bioforce Nano eNabler, showing minimal thickness variations.
considerable distance from the substrate, splashing introduces thickness variations in an un-optimized film (Figure 4.4a). Micro/nanofluidic cantilever arraying tools such as the Bioforce Nano eNabler™ provide an alternative method to selectively pattern nanoliter solutions on various substrates. Contrary to inkjet deposition, the cantilever tip makes direct contact when transferring solution to a substrate (Figure 4.4b). Parameters such as dwell time and contact force are adjustable, enabling precise control over deposited material properties. The Nano eNabler also has built-in capabilities for wafer-level array patterning that can be applied to large-scale sensor functionalization.

Polyethyleneimine (PEI) [69], a hygroscopic (water-attracting) polymer, was selected for humidity sensing based on its high affinity for polar molecules. PEI (50% w/v in H₂O, Sigma-Aldrich) is difficult to selectively pattern for two main reasons: (1) even after dilution to 1% concentration, the solution viscosity is high enough to break the 2-µm thick nitride surface patterning tools (SPTs) or at least clog the narrow channels; and (2) such small amounts of solution quickly evaporate regardless of the deposition environmental conditions. Figure 4.5 depicts the unfortunate manifestation of both issues at once: clogged channels and evaporated solvent (water in this case) resulted in platforms becoming attached to the SPT by dried PEI. To overcome the first challenge,
SPT tips were “front-loaded” with solution rather than relying on capillary forces to move polymer from the SPT reservoir through the channel. Using a micro-centrifuge to mix a small amount of low-vapor-pressure alkaline buffer (Bioforce Nanosciences, Inc.) with PEI resulted in a significantly delayed solution evaporation time. Ultimately, ZnO-on-Si sensors were functionalized by loading a 1:10 mixture of PEI:buffer onto an SPT and depositing the solution onto each platform (Figure 4.6a).

4.2.4 SENSOR MEASUREMENT AND CHARACTERIZATION

Resonator frequency response was characterized using an Agilent E8364B Vector Network Analyzer. The reference sensor frequency $f_{1,\text{ref}}$ and third-order harmonic mode $f_{3,\text{ref}}$ (35 and 86 MHz, respectively) had a measured $Q$ of 1,300 and 2,000 in air, and the RH sensor $f_{1,RH}$ had a $Q$ of 1,340 in air after PEI patterning (Figure 4.6b).

Sensors were simultaneously characterized in a Tenney BTRC environmental chamber; to obtain the reference sensor characteristic for self-temperature sensing, the chamber temperature was varied from 25 to 85°C in 15°C intervals at constant RH values. To obtain the RH sensor behavior, RH values were ramped in 20% intervals from 20 to 92%RH at constant temperature values. Network analyzer measurements of each sensor $f_i$ and $f_3$ were recorded after one hour of stabilization.

Figure 4.6: (a) Optical image of ZnO-on-Si humidity sensor functionalized with PEI; (b) frequency response of PEI-patterned RH sensor; $f_{1,RH} = 37$ MHz, $Q = 1340$ in air.
Equation (7) models the RH sensor total frequency change $\Delta f_{RH}$ as a function of temperature $T$ and $RH$. Because both temperature and RH sensors are structurally identical, they are assumed to have the same temperature response. Thus, the temperature-induced frequency change $\Delta f_{ref}(T)$ is used in place of $\Delta f_{RH}(T)$. $\Delta f_{ref}(T)$ and the total $\Delta f_{RH}$ are experimentally measured to thermally compensate the RH response $\Delta f_{RH}(RH)$.

$$\Delta f_{RH} = \Delta f_{ref}(T) + \Delta f_{RH}(T)$$  \hspace{1cm} (7)$$

$\Delta f_{ref}(T)$ is characterized by a combination of $f_1$ and $f_3$ denoted as beat frequency $f_b = 3*f_1 - f_3$. Equation (8), extracted from the reference sensor response over temperature $T$ ($^\circ$C) at constant RH (Figure 4.7), was used to determine $TCf_b$.

$$\Delta f_{b,ref}(T) = -8.8 \times 10^{-4} \cdot T + 19.35[\text{MHz}]$$  \hspace{1cm} (8)$$

$TCf_b$ is greater than $TCf_1$ and linear, providing greater temperature accuracy and more

![Figure 4.7: Temperature response of $f_{b,ref}$ at constant RH.](image-url)
convenient future calibration. Moreover, $f_{b,\text{ref}}$ shows negligible sensitivity to humidity compared to $f_{b,\text{RH}}$, which was confirmed by sweeping RH at constant temperature values to determine the humidity sensitivity $S_{b,\text{RH}}$, or $\Delta f_b/\Delta \text{RH}$ (Figure 4.8).

The uncompensated RH sensor response exhibits distinguishable shifts with temperature (Figure 4.10). To obtain a compensated RH sensor response (Figure 4.9), the $f_i$ vs. RH characteristics at various temperatures were plotted using thermally-corrected $f_i$ values, confirming that the reference sensor minimizes temperature-induced changes in $f_i$. The average room temperature sensitivity is 1.2 kHz/%RH from 20 to 60 %RH and 14 kHz/%RH from 60 to 92 %RH. The sensor has a measured average hysteresis of 1.8 %RH at 55°C, which is the mid-point of the tested temperature range.

![Figure 4.8: RH sensitivity of beat frequency $f_{b,\text{ref}}$ and $f_{b,\text{RH}}$ over temperature.](image_url)
Figure 4.9: Sensor response after thermal compensation.

Figure 4.10: Sensor $f_i$ response to RH prior to thermal compensation.
The RH sensor response to a rapid increase from 20 to 40 %RH was measured at 25°C (room temperature) (Figure 4.11). $f_{1,RH}$ tracks very closely with RH and has a response time of less than 25 seconds, which is limited by measurement setup and network analyzer sweep time.

![Graph showing sensor response to RH increase from 20 to 40 %RH at 25°C.](image)

**Figure 4.11**: Sensor response to RH increase from 20 to 40 %RH at 25°C.

In this section, the highest $Q$ humidity-sensing BAW micro-resonator was presented [70-72]. The selectively-patterned sensor was thermally compensated from 20 to 92 %RH by measuring the local temperature using an identical device that does not drastically increase the overall footprint. The PEI layer responded to humidity over the complete compensated range, but the RH sensor has an increased sensitivity of 14 kHz/%RH from 60 to 92 %RH. The RH sensor sensitivity at room temperature is on the same order of magnitude as previous works [65, 70, 72] but combined temperature and humidity sensors offer a size reduction compared to other micromachined resonant solutions without a temperature sensor [70]. While the RH&T sensors in this work were separate devices, dual-mode temperature sensing ultimately enables a single device to measure {Pierce, 1998 #218]
Combustion-based sensors have often been used to detect VOCs that readily form explosive mixtures in air; however, they are mainly suitable for concentrations well above permissible exposure limits [73] and require additional power to heat a filament that oxidizes the gas mixture. On the other hand, piezoelectrically-transduced, silicon-based resonant sensors can be easily functionalized for various targets and combined with integrated electronics to create a low-cost, portable air quality detection system for temperature, humidity, VOC and gas sensing.

### 4.3.1 SENSOR DESIGN AND OPTIMIZATION

With $W = L$, the ZnO-on-Si RH sensor utilized an “out-of-phase” extensional mode, which provided lower mass sensitivity than the undetected in-phase mode. Longitudinal extension of the central block (whether $W > L$ or $L > W$) is essential to creating a prominent in-phase sensing mode. An AlN-on-Si resonant sensor with $L = 168$ µm and $W = 80$ µm exhibits significant enhancement of the sensing mode (Figure 4.12).

![Figure 4.12: Frequency response measurements of 16-MHz sensor: (left) optimized sensing mode with platforms and central block in phase and (right) lower sensitivity mode with platforms out of phase with central block.](image)

\[
\begin{align*}
\text{In-Phase (higher mass sensitivity)} & \quad \text{Out-of-Phase (lower mass sensitivity)} \\
\hline
f &= 16.7 \text{ MHz} \\
Q (\text{air}) &= 2540 \\
R_m &= 10.9 \text{ kΩ} \\
f &= 31.1 \text{ MHz} \\
Q (\text{air}) &= 714 \\
R_m &= 35.4 \text{ kΩ}
\end{align*}
\]
Mass sensitivity of the 16-MHz sensor was characterized by depositing platinum (Pt) via focused ion beam (FIB) on each platform and measuring the frequency after each deposition (Figure 4.13). The sensor maintained a $Q$ of 2300 in air and exhibited a linear behavior that agrees well with simulated results. The varying purity of deposited Pt was attributed to discrepancies between calculated and actual mass [26].

In practical sensor implementation, the resonator is connected in feedback with a transimpedance amplifier (TIA) to generate oscillations at $f_0$. The Barkhausen criterion for sustained oscillations indicates that resonator motional resistance $R_m$ and electrical phase directly impact TIA design. Because piezoelectric resonators can excite several modes by a single electrode configuration, gain and phase conditions are often met for many frequencies. Therefore, both adequate compensation of resonator electrical loading [74] and spurious mode suppression are crucial in safeguarding the circuit from locking into oscillations at undesired frequencies.

Figure 4.13: Simulated and measured mass sensitivity of 16-MHz sensor ($L = 168$ µm, $W = 80$ µm, $L_p = 40$ µm); measured data for 0 pg, 368 pg, and 737 pg of mass loading; (inset left) measured sensor frequency response; (inset right) close-up SEM image of FIB-deposited Pt on annexed platform.
Increasing $W$ provides a larger electrode area to reduce $R_m$ without significantly affecting resonance frequency. The resulting $W/L$ ratio should be greater or less than 1 (but not equal) to preserve extensional behavior of the central block. A 19-MHz sensor with 2× larger area ($L = 156 \, \mu\text{m}$, $W = 206 \, \mu\text{m}$, $L_p = 30 \, \mu\text{m}$), which is designed and simulated in ANSYS, shows that lower $R_m$ is achievable without sacrificing PE mode mass sensitivity (Figure 4.14).

![Simulated mass sensitivity of 19-MHz sensor (L = 156 μm, W = 206 μm, L_p = 30 μm) with reduced $R_m$ compared to 16-MHz sensor.](image)

**Figure 4.14:** Simulated mass sensitivity of 19-MHz sensor ($L = 156 \, \mu\text{m}$, $W = 206 \, \mu\text{m}$, $L_p = 30 \, \mu\text{m}$) with reduced $R_m$ compared to 16-MHz sensor.

The entire central resonant block undergoes expansion and contraction at $f_0$, implying that electrical phase shift between input and output electrodes will be 0° regardless of placement on the piezoelectric layer. While interdigitated electrodes reduce the number of spurious modes for the 16-MHz sensor, three peaks with up to 10 dB lower IL than the sensor mode $f_0$ may complicate oscillator interfacing (Figure 4.15a). Thus, optimization goals for the 19-MHz sensor were to reduce $R_m$ while suppressing undesired modes. Three potentially problematic modes at 29 MHz, 57 MHz, and 84 MHz were identified through ANSYS harmonic analysis. Based on the strain polarities shown in
Figure 4.15b, splitting electrodes symmetrically about the W axis creates 0° phase shift at \( f_0 \) and 180° phase shift for the three modes of concern. Therefore, even if a mode exhibits comparable IL to \( f_0 \), circuitry designed for \( f_0 \) (0° phase) would not satisfy the phase condition to lock into oscillations at another frequency.

The measured \( R_m \) of a fabricated 19-MHz sensor was 1.14 kΩ, corresponding to an IL value 10 dB lower than the majority of spurious modes up to 100 MHz (Figure 4.15b). While predicted problematic modes \( f_1, f_2, \) and \( f_3 \) had IL within 5 dB of \( f_0 \), the designed 180° phase shift ensures the modes will not threaten oscillator interfacing. The remaining spurious peaks, which are located at higher frequencies, can be sufficiently attenuated by employing a simple low-pass filter within the feedback loop.
Figure 4.15: Wide-span frequency response for (a) 16-MHz sensor (inset: optical image); (b) 19-MHz sensor with improved IL and spurious suppression. Peaks within 5 dB IL of $f_0$ have 180° phase shift (inset: optical image of fabricated device).
4.3.2 FUNCTIONALIZATION FOR TOluene AND XYlene SENSING

Prior to functionalization, the uncoated sensor was characterized in an ESPEC environmental chamber and reported a temperature coefficient of frequency (TCF) of -30 ppm/°C at 19 MHz. Since AlN-on-Si thermometers are easily incorporated into the sensor array, the large, linear TCF is useful for accurate monitoring and correction of temperature-induced frequency shift [75] as well as determination of relative humidity levels [76].

To validate sensor performance for VOC detection, the Nano eNabler was used to coat released devices with the appropriate functional material. Polyimide, polyethyleneimine (PEI), and similar polymers have been commonly used for detection of polar compounds (e.g. water vapor) [76]. However, the most frequently-appearing indoor chemical pollutants [77] are hydrocarbons consisting of non-polar molecules. Sorption-based resonant VOC sensing has employed polymers such as polyisobutylene (PIB), polydimethylsiloxane (PDMS) [78], and photoresist [79], which exhibit a strong affinity for non-polar compounds. In this work, PIB was selected based on favorable sorption characteristics for toluene and xylene.

PIB, which is commercially available in solid form (Sigma-Aldrich), is typically dissolved using organic solvents such as toluene or xylene. Since the Nano eNabler is designed for biological functionalization, deposition conditions can be controlled better for aqueous solutions. Nanoliter volumes of highly volatile substances (at room temperature, toluene vapor pressure = 2.92 kPa, xylene vapor pressure = 1.33 kPa) are likely to evaporate within seconds during deposition. In the same way that polymer patterning in section 4.2.3 required mixing PEI with a lower-vapor-pressure solvent, PIB
solution requires a second solvent such as diethyl phthalate (DEP), which has been used to facilitate non-polar polymer printing [80] (vapor pressure 0.22 Pa), to prevent evaporation. Trials for various PIB/toluene concentrations and ratios of PIB/toluene to DEP revealed that mixing as little as 0.1% DEP with a 2% PIB/toluene solution was optimal for complete polymer transfer onto annexed platforms without requiring multiple depositions to achieve uniformly thick PIB films (Figure 4.16).

The sensor functionalization process (Figures 4.17b-d) began with preparation of a 2% PIB/toluene solution, which was subsequently diluted in DEP. An SPT loaded with the resulting solution was used to deposit PIB by contacting the SPT to each platform. Following deposition, sensors were baked on a hotplate at 125 °C for two hours to remove any remaining solvent. LEXT confocal imaging of the 1-µm thick PIB patterns revealed a slight dome-shaped profile. Mass loading from the PIB causes $f_0$ to decrease after patterning; however, the initial $Q$ of 3400 increases slightly to 3700 (Figure 4.18a).

Figure 4.16: Optical images of Nano eNabler patterning trials with various PIB solutions: (a-c) 0.1% PIB/xylene solution with no DEP added, showing quick evaporation of a micro-pipetted drop in 3 minutes; (d-f) 1:1 ratio of 2% PIB/toluene to DEP. Optical images over a 10-minute time period show very little polymer remaining upon DEP evaporation.
Figure 4.17: (a) SEM image of unpatterned AlN-on-Si sensor; (b) optical image of sensor during PIB patterning (SPT shown above right platform); (c) optical image of PIB-patterned sensor; (d) Olympus LEXT thickness profile of PIB on platforms (z-axis exaggerated to clearly illustrate PIB pattern shape).

Figure 4.18: Measured frequency response before and after PIB patterning; (b) photo of customized test board for VOC sensor testing.
4.3.3 Experimental Setup and Sensor Measurement Results

Professor Oliver Brand provided equipment and laboratory space for VOC sensor characterization. Patterned devices were wirebonded to a 28-pin DIP chip carrier, which was inserted into a test board containing SMA connections to interface with an HP 4395A network analyzer (Figure 4.18b). The gas mixing setup described in [81] was used to generate three-minute exposures to various toluene and xylene concentrations, which were each followed by a three-minute nitrogen purge to refresh the PIB. Based on the saturation concentration of each VOC at room temperature and atmospheric pressure, the VOC concentration delivered to the sensor was controlled by adjusting flow ratios between N₂ and analyte lines (total flow rate = 80 mL/min). Frequency measurements were acquired at 15-second intervals and plotted over time (Figures 4.19 and 4.21). In all VOC tests conducted, the sensor response to toluene and xylene was reversible and repeatable after briefly exposing the PIB film to analyte-free air.

Toluene testing revealed a linear sensitivity of -0.09 Hz/ppm from 3600 to 14400 ppm toluene. To confirm that the observed frequency change was primarily due to mass loading in the PIB film, an uncoated sensor was measured upon exposure to 20600 ppm toluene, where a 224 Hz shift, nearly 10× lower than the PIB-patterned sensor, was measured. The $Q₀$ of 3700 was unaffected by subsequent mass loading from toluene exposure, exhibiting less than 10% reduction up to 14400 ppm toluene. In comparison, a 29-MHz mode from the same device ($f_i$) demonstrated a less linear sensing behavior. The initial $Q_i$ of 10,500 was reduced to 8500 after PIB patterning and degraded by nearly 50% at a toluene concentration of 14400 ppm (Figure 4.20).
Since the saturation concentration of xylene is approximately three times lower than that of toluene at room temperature and atmospheric pressure, lower xylene concentrations were achievable using the same setup. Additionally, the xylene partition coefficient (ratio of analyte concentration in the polymer to the concentration in the surrounding environment) is two times larger than the toluene partition coefficient [38].

Figure 4.19: Measured frequency change for toluene concentrations of 20600 ppm, 14400 ppm, 7200 ppm, and 3600 ppm.

Figure 4.20: Measured change in $Q$ vs. toluene concentration for sensing mode $f_0$ compared to 29-MHz mode $f_1$; arrows denote platform motion.

Since the saturation concentration of xylene is approximately three times lower than that of toluene at room temperature and atmospheric pressure, lower xylene concentrations were achievable using the same setup. Additionally, the xylene partition coefficient (ratio of analyte concentration in the polymer to the concentration in the surrounding environment) is two times larger than the toluene partition coefficient [38].
Therefore, xylene sorption is expected to be about twice as large as toluene sorption. Measurements revealed a linear sensitivity of -0.18 Hz/ppm from 625 to 3750 ppm xylene (Figure 4.21). Similar to toluene exposure results, $Q_0$ was minimally affected while $Q_1$ severely degraded with mass loading from xylene sorption in the PIB film (Figure 4.22).

![Figure 4.21: Measured frequency change for xylene concentrations of 6250 ppm, 5000 ppm, 3750 ppm, 2500 ppm, and 1250 ppm.](image1)

![Figure 4.22: Measured change in $Q$ vs. xylene concentration for $f_0$ and $f_1$ (for comparison).](image2)
4.3.4 **Humidity Effects on VOC Measurements**

Because true ambient air contains a mixture of gases rather than pure substances, effects of water vapor in the sensing environment were investigated by delivering moist air mixed with N₂ and toluene to the sensor. Relative humidity (RH) contributed to an additional downshift in frequency (Figure 4.23); however, sensor measurements with moist air (but no analyte) revealed that \( Q_0 \) and \( Q_1 \) exhibited less than 5% degradation.

![Graph showing frequency change vs. time for 14400 ppm Toluene](image)

**Figure 4.23:** Measured frequency change for xylene concentrations of 6250 ppm, 5000 ppm, 3750 ppm, 2500 ppm, and 1250 ppm.

![Graph showing change in Q vs. RH for f₀ and f₁](image)

<table>
<thead>
<tr>
<th>%RH</th>
<th>( df/f_0 ) [%]</th>
<th>( dQ/Q_0 ) [%]</th>
<th>( df/f_1 ) [%]</th>
<th>( dQ/Q_1 ) [%]</th>
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<td>0</td>
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<td>0</td>
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<td>-0.0014</td>
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<td>-0.0047</td>
<td>-4.75</td>
<td>-0.0032</td>
<td>-4.06</td>
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</table>

**Table 4.1:** Change in frequency and \( Q \) vs. RH.

**Figure 4.24:** Measured change in \( Q \) vs. relative humidity (RH) for \( f_0 \) and \( f_1 \) of PIB-coated sensor.
from 0% to 82% RH (Figure 4.24, results summarized in Table 4.1). Considering the significant $Q_1$ degradation upon mass loading due to toluene and xylene (Figures 4.20 and 4.22), it is reasonable to conclude that water vapor could not have been adsorbed by the PIB. Because this behavior is expected based on the non-polar affinity of PIB, the additional downshift in frequency is most likely related to surface adsorption onto the AlN film [82]. In future implementations, surface treatment on non-designated sensing areas can help prevent undesired adsorption for improved analyte selectivity.

Table 4.2 summarizes properties of the AlN-on-silicon sensor in this work along with other resonant toluene sensors in literature for fair comparison. This sensor demonstrates a high $Q$ that remains unaffected by mass loading, therefore minimizing effects of undesired frequency error. Additionally, the low $R_m$ (nearly three orders of magnitude lower than capacitive sensors [83]) facilitates low-power oscillator implementation. Sensitivity can be improved by selecting or engineering a material with optimal sorption properties for the target analyte as well as designing the resonator with enhanced gravimetric sensitivity [58].

The studied effects of temperature and humidity on sensor response further confirm the need for integrated sensing. Since the die area of 200 x 200 µm is significantly smaller than typical QCM, SAW, and cantilever sensors, large-scale resonator arrays can be practically implemented in a multi-sensing platform.
Table 4.2: Comparison of resonant toluene sensors in literature.

<table>
<thead>
<tr>
<th>Ref.</th>
<th>Type/actuation method</th>
<th>$f$ [MHz]</th>
<th>$Q$</th>
<th>$R_m$ [kΩ]</th>
<th>Sensitivity [ppm$<em>{freq}$/ppm$</em>{toluene}$]</th>
<th>Functional material</th>
<th>Dimensions</th>
</tr>
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<tr>
<td>This work</td>
<td>AlN-on-Si</td>
<td>19</td>
<td>3700</td>
<td>1.1</td>
<td>.005 (3600 – 14400 ppm)</td>
<td>PIB</td>
<td>186 x 206 µm</td>
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<tr>
<td>Johnston</td>
<td>FBAR (ZnO)</td>
<td>1440</td>
<td>170</td>
<td>-</td>
<td>.005 (1000 – 5200 ppm)</td>
<td>PDMS</td>
<td>100 x 100 µm</td>
</tr>
<tr>
<td>[78]</td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>Debéda [84]</td>
<td>Cantilever (PZT)</td>
<td>0.07</td>
<td>142</td>
<td>-</td>
<td>.2 (260 – 1050 ppm)</td>
<td>PEUT</td>
<td>8 x 2 mm</td>
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<tr>
<td>Konno [83]</td>
<td>Capacitive (Si)</td>
<td>5</td>
<td>1826</td>
<td>750</td>
<td>.012 (1000 – 2000 ppm)</td>
<td>PBD</td>
<td>100 x 100 µm</td>
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<tr>
<td>Sayago [85]</td>
<td>SAW (ST-X quartz)</td>
<td>158</td>
<td>-</td>
<td>-</td>
<td>.02 (25 – 200 ppm)</td>
<td>PIB</td>
<td>mm</td>
</tr>
<tr>
<td>Truax [38]</td>
<td>Disk (Si, thermal)</td>
<td>0.4</td>
<td>2600</td>
<td>-</td>
<td>0.1 (2000-16000 ppm)</td>
<td>PIB</td>
<td>400 x 340 µm</td>
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<tr>
<td>Si [86]</td>
<td>QCM</td>
<td>10</td>
<td>-</td>
<td>-</td>
<td>.016 (200 – 1000 ppm)</td>
<td>P3HT</td>
<td>mm diameter</td>
</tr>
</tbody>
</table>

*Polymer functional material acronyms: polyisobutylene (PIB), polydimethylsiloxane (PDMS), polyetherurethane (PEUT), polybutadiene (PBD), poly(3-hexylthiophene) (P3HT).
5 TPOS DUAL-MODE RESONATORS AND OSCILLATORS FOR TEMPERATURE MEASUREMENT

Microelectromechanical system (MEMS) oscillators based on single-crystal silicon resonators are especially promising candidates for integrated timing and sensing applications. However, achieving a highly-stable oscillator output is challenging when the frequency of silicon-based devices varies with temperature by approximately -30 ppm/°C. Passive techniques such as engineering device geometry [61] and degenerate doping [87] have reduced the silicon TCF to as low as –3.6 ppm/°C and –1.5 ppm/°C, respectively. Traditional methods using a layer of positive-TCF oxide to counteract the negative TCF of silicon [88] are susceptible to process variations and may also degrade the resonator $Q$. Rather than depositing surface oxide, embedding oxide pillars inside the silicon structure completely eliminates the first-order TCF of high-$Q$ AlN-on-silicon resonators, resulting in just 90-180 ppm frequency drift over 120°C [59]. Even with significant advances in material compensation, electronic tuning is still necessary to achieve zero-TCF oscillators. Active temperature compensation has been demonstrated using MEMS-based oscillators [89-91], but investigation of thermometry methods for silicon-based oscillators are beneficial to both timing and sensing applications.

Accurate temperature measurement requires a sensor to be in close proximity to the oscillator or sensor system, thereby reducing discrepancies caused by thermal gradients. Compensating for temperature-induced frequency shift using the resonator itself has been explored in quartz oscillators, since frequency can be measured with high accuracy. Dual-mode (DM) excitation, which exploits the dependence of frequency on physical properties of the resonator, has been utilized to simultaneously monitor multiple
quantities (e.g. stress, mass, pressure, or temperature [92-94]). Tracking internal resonator temperature via DM operation provides a local measurement without external thermometers, which significantly reduces overall system footprint when integrated with CMOS electronics. While quartz-based DM oscillators have been heavily researched over the past few decades, the DM concept has only recently been demonstrated in silicon-based MEMS oscillators [75].

5.1 DUAL-MODE TEMPERATURE MEASUREMENT

5.1.1 PRINCIPLE OF OPERATION

Dual-mode temperature sensing requires simultaneous excitation of two resonance modes in a single device. Both modes experience identical changes in ambient temperature but have unique frequency dependence on temperature [16]. The frequency-temperature characteristics of $f_1$ and $f_2$ can be used to compensate reference frequency $f_1$ over temperature. The DM oscillator output, beat frequency $f_b$, is defined by a linear combination of the two modes:

$$f_b = \alpha \cdot f_1 - f_2.$$  \hspace{1cm} (9)

Multiplicative constant $\alpha$ is an integer when $f_2$ is an $\alpha$-order harmonic of $f_1$ [16]. If the modes are close in frequency, $f_1$ and $f_2$ are directly mixed to generate $f_b$ [95].

The relative change in frequency $f_0$ with respect to temperature, or $TC_{f_0}$, can be expressed as

$$TC^n f_0 = \frac{1}{f_0} \frac{\partial^n f_0}{\partial T^n}. \hspace{1cm} (10)$$

Since silicon has a dominant first-order temperature coefficient, $TC F$ refers to $TC^1 F$ here. Based on (9) and (10), DM temperature sensitivity $TC_{f_0}$ is expressed as
As \( f_b \) approaches a minimum value, \( \alpha f_i \equiv f_2 \), which relates \( \text{TC}_f \) to \( f_2, f_b \), and the difference in TCF between modes (\( \text{TC}_f - \text{TC}_f \), or \( \Delta \text{TCF} \)):

\[
\text{TC}_f \approx \frac{f_2}{f_b} (\text{TC}_f - \text{TC}_f).
\]

The linear TCF of silicon ensures that \( \text{TC}_f \) would also be a strongly linear function of temperature, which enables compensation of reference oscillator frequency \( f_i \) over temperature.

### 5.1.2 Dual-Mode Oscillator Considerations

Real-time measurement and compensation requires circuitry to drive and sustain oscillations across a wide temperature range. Various architectures have been used to implement quartz DM oscillators, which can be categorized as either single-loop or dual-loop configuration [95, 96]. Single-loop (Colpitts) oscillators (Figure 5.1a) require fewer components than a dual-loop oscillator and can exploit transistor nonlinearities to inherently provide \( f_b \) as an output [95, 96].

Dual-loop DM oscillators (Figure 5.1b) can be implemented if separate gain and phase control are required. The Barkhausen stability criterion, which states that a circuit will sustain oscillations under certain loop gain and phase conditions, must be satisfied at both frequencies in a DM oscillator. Unity loop gain is achieved by ensuring that the transimpedance amplifier (TIA) provides sufficient gain to compensate resonator motional resistance (\( R_m \)). Phase conditions are met if the combined TIA and resonator phase results in zero loop phase shift.
In addition to resonator phase and motional resistance $R_m$, $Q$ of both modes directly affects oscillator phase noise and consequently, temperature measurement linearity [75]. Careful design of resonators with proper phase, low $R_m$, and high $Q$ ensures successful DM oscillator interfacing.

Figure 5.1: (a) Single-loop Colpitts oscillator with one-port resonator; (b) dual-loop DM oscillator with two-port MEMS resonator in feedback with TIA's followed by mixing, multiplication, and filtering blocks to generate output frequency $f_b$. 

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5.1.3 TPoS Dual-Mode Resonators

Quartz-based DM stress and temperature measurement exploited a temperature-compensated C-mode and an uncompensated B-mode with TCF ~ -26 ppm/°C [92-94]. Although two modes with such different temperature characteristics are not readily available in silicon, a similar micromachined thermometry concept has been employed in [64]. The dual-resonator sensor utilized a surface oxide-compensated resonator and uncompensated resonator to obtain a large ΔTCF between the two modes, and small $f_b$ was obtained by designing $f_2$ close to $f_1$. The dual-resonator thermometer was more sensitive than single-mode temperature sensing, but using two devices increases die area and may cause measurement inaccuracies. A self-temperature sensing approach using Lamé and square-extensional modes in [97] provides a single-resonator alternative. However, both [64] and [97] are based on electrostatically-transduced (capacitive) resonators, which have high motional impedance and create difficulties in oscillator implementation. The required DC polarization voltage ($V_p$) further increases power consumption and may contribute to undesired measurement inaccuracies via temperature-induced $V_p$ variations.

On the other hand, piezoelectric actuation enables multiple-mode excitation in a single device without DC voltages. Thin film piezoelectric-on-substrate (TPoS) technology combines the advantages of piezoelectric transduction with a low-loss material such as silicon, which exhibits high-$Q$ and enhanced power handling over piezoelectric-only resonators [39].
5.2 ZNO-ON-SI DUAL-MODE OSCILLATOR

A dual-frequency oscillator in [98] was built from a TPoS resonator with discrete circuitry designed to switch between modes. To eliminate manual switching, we present a DM oscillator comprising a ZnO-on-Si resonator interfaced with custom-designed integrated circuits (ICs), which demonstrates the first micromachined alternative to DM quartz thickness-shear “C” mode oscillators [16, 96].

5.2.1 SYSTEM ARCHITECTURE

The ZnO-on-Si DM oscillator simultaneously excites fundamental and third-order length-extensional frequencies ($f_1$ and $f_3$) by the architecture shown in Figure 5.2. The resonator is connected in feedback with a 0° TIA, which locks into oscillations at $f_1$, and a 180° TIA that locks into oscillations at $f_3$. The resistive divider comprising $R_1$ and $R_2$ isolates oscillator outputs $f_1$ and $f_2$ to minimize loading effects. In a complete DM interface and readout system, filters outside the feedback loops remove unwanted signal components at each amplifier output. $f_1$ is then multiplied and mixed with $f_3$, after which the signal is filtered again to extract $f_b$ ($f_b = 3f_1 - f_3$). While all system blocks would

Figure 5.2: Block diagram of dual-mode (DM) interface and readout architecture for ZnO-on-Si BAW resonator.
ultimately be integrated on-chip, measurements for this work were performed via
spectrum analyzer and data post-processing to extract \( f_b \).

Expressing (9) as \( f_b = 3f_1 - f_3 \) to describe the selected modes, (11) becomes

\[
TCf_b \approx \frac{3f_1TCf_1 - f_3TCf_3}{f_b}. \tag{13}
\]

If \( 3f_1 \approx f_3 \) as in [16], (13) is approximated by

\[
TCf_b \approx \frac{f_3}{f_b} (TCf_1 - TCF_3), \tag{14}
\]

highlighting the strong dependence of \( TCf_b \) on TCF difference between first and third
modes (\( TCf_1 - TCF_3 \), or \( \DeltaTCF \)). Additionally, the ratio \( 3f_1/f_b \) can be adjusted to enhance
\( TCf_b \).

5.2.2 Dual-Mode Resonator Design

Fundamental and third-order length-extensional modes (\( f_1 \) and \( f_3 \)) can be sensed
using an interdigitated electrode configuration. Matching input and output electrodes to
alternating strain fields of the third-order mode creates a 180° electrical phase difference
between resonator input and output at \( f_3 \) (Figure 5.3). Since the entire device undergoes
uniform expansion and contraction at \( f_1 \), no phase change occurs between input and
output. The 180° phase difference between modes ensures that each resonance mode is

![Figure 5.3: ZnO-on-Si BAW mode shapes at \( f_1 \) and \( f_3 \).](image)
isolated from the other and enables individual tuning to produce an optimized signal at the DM oscillator output that meets the Barkhausen criterion for sustained oscillations [95]. The measured frequency response of $f_1$ and $f_3$ using the proposed electrode configuration is shown in Figure 5.4.

5.2.3 INTERFACE CIRCUITRY

Each feedback loop in the DM oscillator contains two amplifiers. A front-end TIA (AMI 0.6 µm) provides the required phase shift to lock into oscillations at both frequencies – 0° for $f_1$ and 180° for $f_3$. A discrete buffer (TI OPA656) follows each TIA to isolate the resistive voltage divider load from the IC output. The voltage divider sums the two signals and feeds them back to the resonator input.

Each CMOS IC consists of a front-end TIA with an inverting pair connected in feedback by $M_f$ (Figure 5.5). $V_{bias,\varphi}$ (resonator phase $\varphi$ equals 0° or 180°) is separately
adjusted in both feedback loops to prevent the desensitization effect characterized by (15) from pulling the loop out of dual-oscillations. A second amplification stage is added after each TIA to minimize loading effects of parasitic capacitances at the resonator input and output terminals. These amplifiers also provide the additional phase swing needed to create a total shift of $0^\circ$ around the feedback loop. Since the TIA inherently provides $180^\circ$ phase shift, a common source output stage provides an additional $180^\circ$ for the $f_1$ IC. A source-follower amplifier maintains the total loop phase shift at $180^\circ$ in the $f_3$ IC.

To sustain dual oscillations, the loop gain of each signal path is controlled such that $A_{f_3} \sim A_{f_1}$. When a system contains two interfering signals of different amplitudes, the signal with larger amplitude will act on third-order nonlinearities in the amplifier and attenuate the amplifier gain, as given by (15):

$$a_1' = a_1 \left(1 + \frac{3a_3}{2a_1}V_i^2\right)$$

(15)

where $a_1$ represents the small signal gain of the amplifier, $a_3$ the gain compression caused by the interfering signal on the third-order nonlinearity of the amplifier ($a_3/a_1 < 0$), and $V_i$ the amplitude of the interfering signal [99]. If one signal is significantly larger than the other, the resulting attenuation of the feedback amplifier causes the mode with lower
amplitude to drop out of oscillation. Therefore, gain tuning is added to ensure that oscillations are maintained while simultaneously generating both frequencies.

5.2.4 Oscillator Measurement and Characterization

The resonator was independently excited at $f_1$ (30 MHz) and $f_3$ (87.3 MHz) and measured in single-mode (SM) oscillation conditions. Both loops were then simultaneously connected in feedback with the resonator for DM operation. Oscillator functionality was verified by an external measurement setup. SM operation of $f_1$ or $f_3$ produced a sinusoidal wave with single frequency component on the oscilloscope (Figure 5.6) and corresponding peaks (with harmonics of $f_i$) on the spectrum analyzer output (Figure 5.7). DM operation produced an output containing both $f_1$ and $f_3$ (Figure 5.7). The spectrum analyzer showed distinct peaks at $f_1$ and $f_3$ in DM operation, with an additional peak at $f_{\text{mix}} = f_3 - f_1$ (57.3 MHz) (Figure 5.7). This peak resulted from mixing between the two modes across the interface electronics, which was confirmed by examining the spectra in SM operation of $f_1$ or $f_3$ ($f_{\text{mix}}$ only exists in simultaneous excitation).

Figure 5.6: (top) Measured oscillator response for $f_1$ and $f_3$ SM excitation; (bottom left) DM excitation at $f_1$ and $f_3$; (bottom right) DM output signal after averaging.
The phase noise performance of $f_3$ in SM and DM excitation (Figure 5.8) was measured using the Agilent N5500A phase noise test set. The lower output power of during DM operation prevented $f_1$ measurement. In DM operation, a 10-20 dBc degradation between $f = 1$ kHz and ~1 MHz is observed; however, as $f$ approaches 1 MHz, the DM architecture phase noise improves beyond that of SM excitation. Closer examination of $f_3$ reveals the presence of a secondary tone near resonance in DM operation, which was not observed in SM excitation of $f_3$. Amplifier nonlinearity was suspected to cause cross-modulation between $f_1$ and $f_3$ and generate a frequency close to $f_3$, which resulted in a $1/f^2$ phase noise profile (-20 dB/dec) at small offsets and a constant noise floor at large offset [100]. The secondary mode may also cause the analyzer to lock into this mode during measurement, resulting in a sharper drop in phase noise performance than expected.
Following phase noise analysis, the DM oscillator was placed in an ESPEC SH-240 chamber, where the temperature was cycled from -40°C to 100°C. $f_1$ and $f_3$ were measured with an Agilent E4407B spectrum analyzer across temperature. $TCf_1$ and $TCf_3$, the TCF values of each frequency in DM operation, were determined to be -34 ppm/°C and -40 ppm/°C, respectively (Figure 5.9). Extracted $TCf_b$ was +162 ppm/°C, showing an enhancement in temperature sensitivity compared to measuring $f_1$ or $f_3$ separately. Because $f_b$ is fairly large at 3 MHz, $TCf_b$ can be more accurately predicted using (13) rather than (14).

Resonator $Q$ loading in simultaneous DM excitation causes significant jitter (frequency uncertainty) at $f_1$ and $f_3$[75], which directly affects the accuracy of frequency-and ultimately temperature--measurement. To determine the effects of $Q_{f1}$ and $Q_{f3}$ on the output signal linearity, several devices were tested over temperature, after which the correlation coefficient of the temperature measurement ($R^2$) was plotted against $Q_{min}$, the minimum $Q$ of the two operating modes (Figure 5.10). Significant variation exists between TCF measurements for a low $Q_{min}$, resulting in poor linearity and a low $R^2$. 

![Phase noise response in SM and DM configuration](image)

**Figure 5.8:** Measured $f_3$ phase noise response in SM and DM configuration.
However, increasing $Q_{\text{min}}$ improves temperature measurement linearity, suggesting a logarithmic relationship between $Q_{\text{min}}$ and $R^2$. Thus, high $Q$ of both modes is crucial to performing accurate temperature measurements using DM oscillators.

![Figure 5.9: Measured temperature response of (a) $f_1$, $f_3$; (b) $f_b$.](image)

![Figure 5.10: Measurement correlation ($R^2$) vs. $Q_{\text{min}}$.](image)
5.3 OPTIMIZED ALN-ON-SI DUAL-MODE RESONATORS

Based on (12), a large \( f_2 \) combined with small \( f_b \) provides enhanced DM temperature sensitivity (ratio \( f_2/f_b \)). In-plane bulk modes of AlN-on-Si resonators provide lithographically-defined frequencies that can be exploited for DM operation. Further \( \Delta TCF_b \) improvements are achievable by maximizing \( \Delta TCF \) (\( TCF_1 - TCF_2 \)), which is influenced by temperature coefficients of independent elastic constants (\( TC_{c11}, TC_{c12}, \) and \( TC_{c44} \)) and their contributions to the selected modes [101].

5.3.1 LIMITATIONS OF FUNDAMENTAL AND HIGHER-ORDER MODES

DM quartz thermometers based on first and third harmonic C-modes have demonstrated \( f_b \) values in the 150-kHz range \( (f_b = 3f_1 - f_3) \) [16, 95]. In previous DM demonstrations, TPoS resonators employed first and third width-extensional (WE) modes [75, 98], which tend to suffer from larger \( f_b \). One factor causing WE/WE3 resonators to deviate from the ideal situation (\( \alpha = 3 \)) is support size, which creates non-uniformities in the \( f_1 \) strain pattern and shifts the frequency from the nominal value. However, finite element simulations show that even with zero support size, \( f_b \) reaches a lower limit for the given dimensions (Figure 5.11). Since width \( W \) determines frequency, the only variable dimension is length \( L \), which also provides minimal \( f_b \) reduction as measured in fabricated resonators (Figure 5.12).

Because \( f_1 \) and \( f_3 \) have the same elastic dependencies, small \( \Delta TCF \) is also expected. For the devices measured in Figure 5.12, \( \Delta TCF = -0.5 \text{ ppm/}^{\circ}\text{C} \) yields \( TCF_b \sim -37 \text{ ppm/}^{\circ}\text{C} \), which provides virtually no improvement over single-mode measurement. Since extensional modes in TPoS resonators possess similar temperature-frequency behavior and large \( f_b \), alternative modes for \( TCF_b \) enhancement are investigated.
5.3.2 \textbf{TCF}_B \textbf{ENHANCEMENT BY BEAT FREQUENCY REDUCTION}

\textbf{In-Plane Width-Extensional and Width-Shear Modes}

The WE mode frequency is nearly independent of resonator length ($L$) and is defined by

\[ f_{WE} = \frac{v}{2W} \]  \hspace{1cm} (16)
where \( v \) is the effective longitudinal wave velocity. On the other hand, the width-shear (WS) mode (Figure 5.13) exhibits a significant dependence on \( L \), which can be exploited to define specific \( f_b \) values by controlling the distance between WS and WE modes \( (f_b = f_{WS} - f_{WE}) \) (Figure 5.14). Utilizing WS and WE modes for DM operation provides inherently small \( f_b \) by directly mixing two frequencies rather than utilizing fractional multiplication to obtain a desired \( f_b \) [96]. \( L/W = 1.31 \) corresponds to the case when \( f_{WS} = f_{WE} \), or zero \( f_b \). Because \( f_b \) is determined by a ratio rather than absolute dimensions, DM excitation using WS and WE modes is less sensitive to lithography and process variations. Furthermore, the WS/WE resonator electrode configuration provides a 180°

![Figure 5.13: ANSYS mode shapes of width-shear (WS) and WE modes for dual-mode operation.](image)

![Figure 5.14: Simulated frequency of width-shear (WS) and WE modes vs. \( L \) for \( W = 156 \) \( \mu \)m, corresponding to \( f_{WE} = 27 \) MHz.](image)
phase shift between modes when operated in two-port configuration [74], which effectively isolates one mode from the other in a dual-loop oscillator.

$TC_{C11}, TC_{C12},$ and $TC_{C44}$ values from [101] were used to model silicon elastic properties in ANSYS to determine WS and WE mode behavior over temperature. $TC_{f_{WS}} = -31.1$ ppm/$^\circ$C and $TC_{f_{WE}} = -32.5$ ppm/$^\circ$C were predicted by pre-stressed modal analyses. WS/WE resonators with $f_{WE} = 27$ MHz were designed with $f_b$ ranging from 17 kHz to 1.5 MHz.

Resonators were fabricated on a 20-µm thick SOI substrate by the process described in chapter 3 and characterized in an ESPEC temperature chamber. The measured frequency response of several devices, obtained with an Agilent E8364B network analyzer, agreed well with simulated $f_b$ values (Figure 5.15). The slightly lower measured $TC_{f_{WS}}$ and $TC_{f_{WE}}$ values (-30.1 ppm/$^\circ$C and -31.1 ppm/$^\circ$C) can be attributed to variation in doping concentration (simulations assumed boron-doped silicon with 4 Ω-cm resistivity, while fabricated device resistivity ranged from 1 to 20 Ω-cm). The highest extracted $TC_{f_b}$ was 1480 ppm/$^\circ$C for a device with $f_b = 17$ kHz (Figure 5.16), nearly 50×

![Figure 5.15: Measured vs. simulated $f_b$ for WS/WE resonator designs with $W = 156$ µm, $L$ varied to modify $f_b$ values ($f_b = f_{WS} - f_{WE}$); (inset) SEM image of WS/WE resonator ($W = 156$ µm, $L = 204$ µm).]
larger than single-mode measurement of $T_{Cf_{WS}}$ or $T_{Cf_{WE}}$. Table 5.1 summarizes measured results for WS/WE resonators.

**Figure 5.16:** Extracted $T_{Cf_b}$ of WS/WE DM resonator with $f_b = 17$ kHz; measured $T_{Cf_{WS}} = -30.1$ ppm/°C, $T_{Cf_{WE}} = -31.1$ ppm/°C.

**Table 5.1:** Summary of WS/WE mode DM resonator performance.

| $L$ [µm] | $Q_{WS}$ | $T_{Cf_{WS}}$ [ppm/°C] | $Q_{WE}$ | $T_{Cf_{WE}}$ [ppm/°C] | $|f_b|$ [kHz] | $|T_{Cf_b}|$ [ppm/°C] |
|----------|-----------|-------------------------|----------|-------------------------|--------------|-----------------------|
| 204      | 9320      | Sim. -31.0 Meas. -30.1 | 2942     | 3128                    | 17           | 1482                  |
| 205      | 13864     | Sim. -32.5 Meas. -31.1 | 3805     | 5492                    | 48           | 553                   |
| 206      | 14803     | Sim. -32.5 Meas. -31.1 | 5492     | 6448                    | 134          | 243                   |
| 210      | 15428     | Sim. -32.5 Meas. -31.1 | 5492     | 6448                    | 507          | 88                    |
| 222      | 16173     | Sim. -32.5 Meas. -31.1 | 6448     | 6448                    | 1531         | 49                    |

Simulated vs. measured $T_{Cf_{WS}}$ and $T_{Cf_{WE}}$ with extracted $f_b$ and $T_{Cf_b}$ of fabricated 27-MHz WS/WE resonators ($W = 156$ µm).
WS/WE Mode Oscillator Interfacing

Acoustic coupling between modes as a result of non-linear effects [102] can cause significant $Q_{WE}$ degradation when WS and WE modes converge (Figure 5.17). Since $Q$ affects frequency stability through oscillator phase noise, interference between modes must be minimized. Devices with smaller $f_b$ may benefit from one-port excitation using two single-loop oscillators, which effectively isolates one mode from the other (Figure 5.18b). However, mode isolation is also achievable using a three-terminal configuration with common input and separate output electrodes for $f_{WS}$ and $f_{WE}$ (rather than the current layout shown in Figure 5.18a).

![Image of frequency response](image.png)

**Figure 5.17:** Measured response of DM resonators with $|f_b| = 1.5$ MHz and 134 kHz ($|f_b| = |f_{WS} - f_{WE}|$), showing $Q_{WE}$ degradation when $f_{WS}$ and $f_{WE}$ converge.
Figure 5.18: WS/WE DM terminal configurations: (a) dual-loop oscillator with two-port operation; (b) two separate one-port oscillators, which provide better isolation of each mode.
5.3.3 Effect of Orientation on WS/WE Resonator TCF

AlN-on-Si WS/WE resonators provide substantial $f_b$ reduction as well as a 2× larger $\Delta$TCF than the WE/WE3 configuration. However, further $\Delta$TCF enhancement would contribute to increased sensitivity as well as relaxed $f_b$ requirements.

Because Young’s Modulus ($E$) in single-crystal silicon is determined by elastic constants $c_{ij}$ and crystal direction, device orientation influences the temperature-frequency behavior of each mode. While WE-mode resonators on (100) substrates have exploited the [100] direction for nearly full compensation for TCf$_{WE}$ [103], devices in this work are re-oriented to increase WS/WE resonator $\Delta$TCF beyond 1 ppm/$^\circ$C.

[100]-aligned WS/WE resonators had a simulated TCf$_{WS}$ = -30.6 ppm/$^\circ$C and TCf$_{WE}$ = -32.7 ppm/$^\circ$C ($\Delta$TCF = 2.1 ppm/$^\circ$C). Devices were designed by a similar methodology to [110]-aligned devices, yielding a matched $f_b$ ratio $L/W$ of 1.71. In the [100] direction, $W = 156 \ \mu$m corresponds to $f_{WE} = 25 \ \text{MHz}$ ($E_{[100]} < E_{[110]}$).

Fabricated devices had a measured $\Delta$TCF = 1.9, nearly 2× greater than [110]-oriented WS/WE resonators (Figure 5.19). Because $f_{WS}$ is easily driven into nonlinear operation [74], $f_{WE}$ is more likely to be used as a reference and requires lower $R_m$, which is achieved through increased electrode area by the required $L/W$ ratio (Figure 5.20).
Figure 5.19: Measured TC$_{fWS}$ and TC$_{fWE}$ for [100]-aligned WS/WE resonator, exhibiting 2× larger ΔTCF than [110]-aligned devices; (inset) SEM image of [100]-aligned resonator.

Figure 5.20: Measured response of [100]-oriented WS/WE resonator with $f_b = 120$ kHz ($W = 156$ µm, $L = 268$ µm).
5.3.4 Dual-Mode Excitation for Temperature-Stable Beat Frequency

While DM WS/WE mode resonators are applicable to high-sensitivity on-chip thermometry, they can also be employed as temperature-stable references, which are required for counter circuits that would be ultimately used to measure the resonator frequency change with temperature.

Equating the right-hand side of (11) to zero provides the requirements to achieve temperature-insensitive \( f_b \) in DM resonators:

\[
f_2 = \alpha f_1 \frac{T C f_1}{T C f_2}.
\]  \hspace{1cm} (17)

Although \( \alpha f_1 \) and \( f_2 \) have non-zero TCF, \( f_b \) (generated by mixing the two signals) will remain constant over temperature. For a DM resonator with known values of \( T C f_1 \) and \( T C f_2 \), \( \alpha f_1 \) and \( f_2 \) must be designed to satisfy conditions in (17). Because \( f_{WS} \) can be modified to a specific ratio of \( f_{WE} \), WS/WE resonators can provide \( f_b \) that is either temperature-sensitive or temperature-insensitive.

A [100]-aligned WS/WE resonator with \( T C f_{WS} = -29.6 \) ppm/°C and \( T C f_{WE} = -31.5 \) ppm/°C requires \( f_{WS} \) to equal \((1.06) \cdot f_{WE}\). A 25-MHz device fulfilling the aforementioned condition (\( f_{WS} = 26.8 \) MHz, \( f_{WE} = 25.3 \) MHz) was fabricated and characterized over temperature (Figure 5.21). The extracted \( T C f_b \) of 2.4 ppm/°C and \( T C^2 f_b \) of 3.3 ppb/°C² correspond to a frequency drift of 266 ppm from -20°C to 100°C (Figure 5.22), which is lower than degenerately-doped resonators [104] or geometrically-engineered capacitive resonators [105].
Figure 5.21: Measured response for \( f_{WS} \) and \( f_{WE} \) of [100]-aligned resonator with \( f_b = 1.5 \) MHz and near-zero TC\( f_b \) (\( W = 156 \) µm, \( L = 242 \) µm).

\[ Q_{WS} = 15400 \quad \text{TC}_{f_{WS}} = -29.7 \text{ ppm/°C} \]
\[ Q_{WE} = 6830 \quad \text{TC}_{f_{WE}} = -31.5 \text{ ppm/°C} \]

Figure 5.22: Extracted TC\( f_b \) of [100]-aligned WS/WE DM resonator with nearly temperature-stable \( f_b \).

\[ \text{TC}_1 f_b = 2.4 \text{ ppm/°C} \]
\[ \text{TC}_2 f_b = 3.3 \text{ ppb/°C}^2 \]
6 TOWARDS INTEGRATED MICROMECHANICAL KINETIC ENERGY HARVESTING

Readily-available low-frequency mechanical energy, such as human walking at 1-2 Hz [106] and other common vibration sources up to 150 Hz [107], can be collected and transformed into useful electric power. However, because kinetic energy harvesting (KEH) devices require large mass to operate at low frequencies, it is challenging to miniaturize such devices without increasing the operation frequency. For example, piezoelectric cantilever harvesters can generate sufficient power to operate wireless temperature sensors [108], but maximum power is only achievable at the device resonance frequency in hundreds of Hz or greater (Figure 6.1a). Thus, lower-frequency vibrations cannot be converted with the same efficiency.

6.1 MULTI-DEGREE-OF-FREEDOM ENERGY HARVESTING

Although environmental vibrations occur across multiple degrees of freedom (multi-DOF), most energy harvesters operate as a single-DOF mass-spring system [109]. To provide power regardless of device orientation with respect to the vibration source, typical devices must be duplicated and oriented along different axes, further increasing system size and complicates integration. A multi-axis device eliminates the need to manually align the harvester for maximum power. Realizing micromachined out-of-plane and in-plane harvesters on the same die is the first step towards achieving this goal.

6.2 MECHANICAL FREQUENCY UPCONVERSION

As electromechanical transducer resonance frequency increases, higher power levels can be captured for constant input vibration amplitudes [109]. However, the
amount of useful ambient energy decreases at higher frequencies. Mechanical frequency upconversion has been explored as a method to combine low-frequency ambient energy scavenging with resonant MEMS devices. Recently reported harvesters utilize an inertial mass that collects environmental energy in a wide frequency range and actuates high-frequency transducers to increase the power output (Figure 6.1b) [110, 111]. However, the overall volume, including external magnets, is in hundreds of mm$^3$ or larger.

### 6.3 ALN-ON-SI MEMS KINETIC ENERGY HARVESTERS

Devices with integrated upconverting transducers are designed to enhance power output while maintaining a small footprint, multi-axis KEH (Figure 6.2). Low-frequency in-plane and out-of-plane vibrations are harvested and upconverted to higher frequencies without post-process assembly or large external transducers.

![Figure 6.1: Schematic diagrams depicting conversion mechanism in previously reported MEMS resonant kinetic energy harvesters: (a) without mechanical frequency upconversion [112]; (b) with mechanical frequency upconversion [113].](image-url)
6.3.1 **OUT-OF-PLANE ENERGY HARVESTER**

The out-of-plane harvester comprises a seismic mass tethered to the substrate with integrated AlN-on-Si beam transducers. The mass captures low-frequency ambient vibrations in an out-of-plane translational mode (Figure 6.3, top). Additionally, the mass motion couples energy into the transducer high-frequency fundamental clamped-clamped beam mode (Figure 6.3, bottom). To sense a high-strain region of the beam deflection, where the most charge is generated by the piezoelectric effect, transducer electrodes and AlN cover the area near the substrate anchor point.

![Figure 6.2: Backside view of a silicon die with in-plane and out-of-plane kinetic energy harvesters fabricated side-by-side.](image)

![Figure 6.3: Out-of-plane harvester mode shapes ($f_{mass} = 149$ Hz and $f_{beam} = 13.7$ kHz) and basic principle of operation.](image)
6.3.2 In-Plane Energy Harvester

Rather than orienting out-of-plane harvesters to detect in-plane vibrations, a second integrated design is presented. The in-plane KEH device consists of a seismic mass with free-standing cantilever micro-picks (μ-picks) located in the center of the mass (Figure 6.4a). The mass exhibits translational motion in response to low-frequency vibrations (Figure 6.4b). When external acceleration generates sufficient force, each μ-pick “snaps” an AlN-on-Si spring transducer that subsequently vibrates at higher frequencies (Figure 6.4c). The spring transducer top electrode connects regions of identical strain polarity to collect charge when kHz in-plane modes are excited by the μ-picks (Figure 6.5).

Figure 6.4: Conceptual diagram of in-plane harvester: (a) rest position, (b) side view of low-frequency seismic mass vibration, (c) side view of high-frequency spring transducer vibration.
The TPoS fabrication process lends itself well to realizing low-frequency energy harvesting devices (10 – 100s of Hz) and high-frequency resonators on the same substrate. AlN-on-Si KEH devices were fabricated a 5-μm device SOI wafer using the process described in section 3.2.3, with minor modifications to provide the necessary mass for low-frequency operation and the capability for in-plane vibration harvesting (Figure 6.6). First, rather than completely remove the handle Si, the backside oxide mask was used to create the seismic mass, connected to the device layer through the buried oxide. Immediately after backside cavity patterning, an additional mask layer was used to selectively etch the backside oxide and control seismic mass thickness, which enables any number of devices on the same wafer to be customized for a particular operating frequency or power requirement. Secondly, topside HF release formed z-shock stops to ensure that in-plane harvesters maintain μ-pick alignment (Figure 6.7). Immediately following HF release, devices were placed in a supercritical dryer to prevent shock stops from adhering to the handle layer through stiction forces.
Figure 6.6: Cross-section view of TPoS sensors and energy harvesters on SOI substrate.

Figure 6.7: Static structural simulation showing 260-µm thick seismic mass deflection resulting from 1g gravitational acceleration, demonstrating the need for shock stops to maintain in-plane alignment with μ-picks.
Figure 6.8: SEM images of out-of-plane harvester. AlN and top Mo are patterned to sense the fundamental clamped-clamped beam mode, exposing bottom Mo in the beam center.

Figure 6.9: (a) SEM image of in-plane harvester z-shock stop removed to show seismic mass formed by handle Si; (b) SEM image of intact shock stop resting on handle Si (anchored to substrate); (c) SEM image of µ-picks and spring transducer; (d) backside optical view of entire device.
6.5 KEH DEVICE CHARACTERIZATION

Devices were mounted on a stage that provides sinusoidal out-of-plane or in-plane acceleration $a_{in}$ with frequency $f_{in}$ (Figure 6.10). All four beam transducer outputs from the out-of-plane devices were wirebonded and connected in parallel while a single spring transducer output from the in-plane device was connected for testing (Figure 6.11). Output current was fed through a load resistance $R_L$ to a transimpedance amplifier (TIA). The TIA output voltage, which is proportional to the converted current, was measured by an oscilloscope.

![Diagram of KEH device measurement setup](image)

**Figure 6.10:** Schematic of KEH device measurement setup for out-of-plane harvesters and in-plane harvesters. A vertical shaker table (MB Dynamics, Inc.) was used for out-of-plane testing while an accelerometer setup (built in-house by former colleagues) was utilized for in-plane testing.

![Optical images](image)

**Figure 6.11:** Optical images indicating electrical connections made to testing setup: (a) all top electrodes of the out-of-plane harvester were wirebonded for testing; (b) a single spring transducer of the in-plane harvester was connected.
6.5.1 Out-of-Plane Energy Harvester

The out-of-plane harvester was first characterized over various accelerations at an input frequency close to the seismic mass resonance. A sinusoidal output signal \( f_{\text{out}} = f_{\text{in}} = 126 \) Hz was generated by the harvester, while increasing \( a_{\text{in}} \) produced large transient current spikes occurring with frequency \( f_{\text{spike}} \) in addition to the sinusoidal signal. When \( a_{\text{in}} \) was increased to \( 0.7g \), \( f_{\text{spike}} \) also became equal to \( f_{\text{in}} \) (Figure 6.12).

Figure 6.12: Out-of-plane beam transducer output at \( a_{\text{in}} = 0.3 \) g, \( a_{\text{in}} = 0.5 \) g, and \( a_{\text{in}} = 0.7 \) g.
The output current impulses were found to have a main spectral component at \( f_{\text{ring}} \) 12 kHz (Figure 6.13), which corresponds to the fundamental beam mode. A peak power of 3.23 nW was measured in response to a 134-Hz, 0.6g acceleration input.

![Figure 6.13: Out-of-plane beam transducer output with periodic upconversion spikes; (inset) close-up view of signal ring-down with 12kHz spectral component.](image)

**6.5.2 In-Plane Energy Harvester**

Similar acceleration characterization was performed for the in-plane harvester at lower frequencies (Figure 6.14). Limitations of the test setup caused frequency to change slightly with acceleration amplitude. Preliminary measurements showed that a minimum acceleration of 0.45g was required to induce periodic actuations of the spring transducer with an \( f_{\text{spike}} \) equal to \( f_{\text{in}} \). Missed actuations may result from imperfect \( \mu \)-pick alignment due to slight tilting of the sample.
For $a_{in} = 0.45g$ and $f_{in} = 2$ Hz, complete actuations of one in-plane spring transducer occurred with $f_{spike} = f_{in}$. A single upconversion spike produced a peak power of 2.27 nW (Figure 6.15). Spectral analysis confirmed signal components at $f_{ring} = 2$ kHz and 7 kHz contained in the upconversion spikes, which are in close agreement with simulated spring modes.

Figure 6.14: In-plane spring transducer output at various accelerations ($R_L = 560 \Omega$).
While testing limitations prevented simultaneous measurement of both spring transducers, the total output current should nearly double as a result of the transducer symmetry [111]. The cantilever transducers on the seismic mass, which are also covered with piezoelectric material, can be measured in conjunction with the spring transducer output to maximize power output harvested from the KEH device.

Power calculations were based on the measured output current and $R_L$. While $R_L = 560 \, \Omega$ was used for in-plane acceleration characterization, measurements with $R_L$ values up to 560 kΩ (Figure 6.16) confirmed that the output power can be maximized by selecting $R_L$ to approach the source impedance of the harvester.

This work introduced micromachined multi-axis AlN-on-Si KEHs as a solution to scavenge low-frequency mechanical energy from the environment. Both out-of-plane and
in-plane devices utilized integrated frequency upconversion transducers that increased the power output beyond the already-harvested energy at the tested input frequencies.

The fabrication process accommodates for multiple degrees-of-freedom by incorporating out-of-plane and in-plane harvesters on the same substrate. Additionally, the fabrication process allows for multiple operating frequencies and increased power output by controlling the seismic mass thickness.

Figure 6.16: Measured peak power vs. $R_L$ for in-plane harvester.
7 CONCLUSIONS AND FUTURE DIRECTIONS

This dissertation presented a MEMS-based environmental monitoring platform comprising TPoS gravimetric resonators and kinetic energy harvesters. Resonators with annexed platforms were shown to be suitable candidates for portable integrated gas sensing as pure mass sensors that can be further optimized for high performance and low-power operation.

7.1 TECHNICAL CONTRIBUTIONS

I. Silicon BAW resonators with annexed platforms for gravimetric sensing

A. Optimization and characterization of TPoS gravimetric resonators: the enabling device of the TPoS micro-platform was designed for high $Q$ and low $R_m$ with optimized sensing mode excitation and spurious mode suppression, which are essential for low-power oscillator interfacing and sensor readout. The annexed platform design enables the sensor to maintain high $Q$ upon mass loading.

B. TPoS resonators for RH&T sensing: ZnO-on-Si resonators were designed, fabricated, and selectively functionalized for gravimetric relative humidity measurement. The RH sensor response was thermally corrected using an identical, uncoated resonator to measure temperature, enabling this functionality to be easily realized in an array configuration where devices are fabricated side-by-side. However, dual-mode temperature sensing can eventually be applied to integrated RH&T sensing with a single device.

C. TPoS resonators for VOC sensing: AlN-on-Si sensors were functionalized for toluene and xylene detection, maintaining high $Q$ during exposure. Humidity and
temperature effects on sensor response were also studied, confirming the need for integrated RH&T sensing to calibrate the VOC sensor response.

II. TPoS resonators for dual-mode temperature measurement

A. Simultaneous dual-mode oscillator for self-temperature sensing: the first silicon-based DM oscillator for self-temperature sensing was demonstrated using ZnO-on-Si BAW resonators. The differing TCFs of two modes were exploited to accurately and locally measure device temperature with a 4× greater temperature sensitivity than single-mode measurement. The effects of simultaneous excitation on oscillator phase noise were also characterized and determined to be a strong function of the minimum $Q$ of the two modes.

B. TPoS resonators for enhanced dual-mode temperature sensing: AlN-on-Si resonators exploiting alternative resonance modes for DM operation achieved reduced beat frequency through geometry engineering and increased $\Delta$TCF through device orientation. For reduced-$f_b$ DM resonators based on width-shear and width-extensional modes, nearly 50× greater temperature sensitivity was achieved compared to using fundamental and third-order width-extensional modes for DM operation.

III. TPoS kinetic energy harvesting devices

A. Multi-axis AlN-on-Si kinetic energy harvesters: devices with with fully-integrated upconverting transducers were fabricated by a process that is fully compatible with TPoS resonators and sensors. Out-of-plane and in-plane vibrations were captured by devices with transducers operating in the kHz range, producing peak output power in the nW range.
7.2 FUTURE WORK

While exciting progress has been made towards the goal of a miniaturized multi-sensing platform, various challenges remain for future investigation.

7.2.1 WAFER-LEVEL SENSOR FUNCTIONALIZATION AND OPTIMIZATION

The Nano eNabler proved to be a useful tool for polymer printing and sensor characterization. However, the TPoS platform should ultimately be compatible with batch micromachining processes such as lithographic patterning and stamping. A hybrid serial/parallel approach would provide the benefits of selective patterning while expediting functionalization and achieving better repeatability and uniformity.

Further material characterization is also required to improve sensor performance as well as provide optimal, repeatable functional layers regardless of processing technique. While top-down patterning has been used to functionalize sensors with non-traditional materials, bottom-up techniques have been successfully used in recent years [81]. Thus, a combination of material types could provide unprecedented selectivity and sensitivity improvements.

7.2.2 WAFER-LEVEL PACKAGING FOR TPoS MICRO-PLATFORM

Although TPoS resonant sensors are capable of high-$Q$ operation without vacuum encapsulation, any micro-sensor requires minimal protection from the environment to ensure that dust or other large particles do not interfere with long-term operation. Wafer-level polymer packaging is a low-cost option that can be made compatible with the TPoS process [114]. Of course, sensor performance under packaged conditions should also be taken into consideration.
7.2.3 **Energy Harvester Reliability and Performance Optimization**

The proof-of-concept KEH devices presented in this thesis demonstrated that the TPoS platform can be successfully applied to other devices of importance to portable and wireless systems. However, further understanding is required to address scaling limits and implementation challenges.

**Materials and Methods for Improved Reliability**

Long-term tests were not conducted on µ-pick reliability, but repeated stress on moving parts is expected to contribute to wear and eventual device, especially in-plane devices (Figure 7.1). One option is to use sputtering or atomic layer deposition (ALD) to coat frictional surfaces with materials such as hafnium diboride and titanium nitride, which have been shown to protect surfaces.

While thicker device layer substrates would improve robustness of the transducer springs, the increased operating frequency must be considered in the design phase. A different structural material, rather than increased dimensions and features of current TPoS devices, could provide the compliance needed for lower frequency harvesting.

![Figure 7.1: SEM image of in-plane harvester with mis-aligned shock stop.](image-url)
To Upconvert or Not To Upconvert?

The difference between harvested power from low-frequency vs. upconverted frequency must be thoroughly investigated. Because the maximum harvestable input power from the seismic mass is larger than that of the spring or beam transducers (based on volume), it is important to determine exactly how much benefit, if any, upconversion provides in regards to power enhancement.

7.2.4 Energy Harvester Implementation and Interfacing

Single-Device Multi-Axis Energy Harvesters

Although devices presented in this thesis were optimized for single-axis energy harvesting, multi-axis kinetic energy scavenging capabilities would be highly beneficial for applications where die real estate is crucial. Integrated multi-DOF designs would be a great contribution to the TPoS micro-platform.

Energy Harvester Signal Conditioning Circuitry

In practical implementation, the MEMS device is only part of an entire energy harvesting system (Figure 7.2). Successful energy extraction requires appropriate signal conditioning and rectification circuitry as well as on-chip storage, such as supercapacitors or micro-batteries. While device and electronics integration are facilitated by the TPoS platform, system architectures dependent on the transduction mechanism must be considered to maximize deliverable energy to wireless and autonomous sensor nodes.
Figure 7.2: Schematic diagram of a MEMS energy harvesting system describing the process of harvesting ambient mechanical energy with a micro-mechanical system, converting it to usable electrical energy by various transduction mechanisms, and processing it for storage and eventual delivery.
REFERENCES


